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# CERIUM OXIDE (CeO<sub>2</sub>) PROMOTED OXYGEN CARRIER DEVELOPMENT AND SCALE MODELING STUDY FOR CHEMICAL LOOPING COMBUSTION

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CERIUM OXIDE (CeO<sub>2</sub>) PROMOTED OXYGEN CARRIER DEVELOPMENT  
AND SCALE MODELING STUDY FOR CHEMICAL LOOPING COMBUSTION

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DISSERTATION

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A dissertation submitted in partial fulfillment of the  
requirement for the degree of Doctor of Philosophy in the  
College of Engineering  
at the University of Kentucky

By

Fang Liu

Lexington, Kentucky

Director: Dr. Kozo Saito, Professor of Mechanical Engineering

Lexington, Kentucky

2013

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## ABSTRACT OF DISSERTATION

### CERIUM OXIDE (CeO<sub>2</sub>) PROMOTED OXYGEN CARRIER DEVELOPMENT AND SCALE MODELING STUDY FOR CHEMICAL LOOPING COMBUSTION

According to IPCC reports, the greenhouse gas CO<sub>2</sub> is responsible for global climate change. Studies show that CO<sub>2</sub> concentration reached a level of 400 ppm in 2013, or 40 % above pre-industrial levels. The contribution of CO<sub>2</sub> from industrial activity to increasing global CO<sub>2</sub> concentrations is widely accepted and points to the need to reduce the emission of this greenhouse gas. One possible combustion technology that shows promise for reducing CO<sub>2</sub> emissions is chemical looping combustion (CLC). It is an oxy-fuel technology, but has the advantages of *in situ* oxygen separation, low NO<sub>x</sub> emissions and low cost of CO<sub>2</sub> emission abatement; it entails the use of an oxygen carrier (OC) to provide oxygen for combusting fuels.

OC development is an important task in CLC. Iron based OCs have attracted most research attention in recent years, mainly due to their inexpensive and non-toxic nature. Bi-metal oxide OCs usually impart better CLC performance than mono-metal oxide OCs, one example of which is the introduction of CeO<sub>2</sub> as a partially reducible material capable of generating oxygen vacancies that lead to oxygen storage and transfer. In this study, CeO<sub>2</sub> was used as an additive to a Fe<sub>2</sub>O<sub>3</sub>-based OC and its effect on physical properties, such as morphology, surface area and mechanical strength, was analyzed in detail. The reactivity of OCs is studied using TGA-MS and a bench scale CLC setup. The results show that the reduction reaction at the surface is independent of whether CeO<sub>2</sub> is present or not, but after the surface oxygen had been consumed, the OC with CeO<sub>2</sub> provided faster oxygen transfer rates from the bulk to the surface to produce better average reaction rates. The OCs after reduction and oxidation were analyzed using XRD and Raman spectroscopy; based on these analytical data, a model for the promoting role of CeO<sub>2</sub> is discussed. Furthermore, the reaction kinetics of the OCs were also studied using shrinking core model, the kinetics parameters were obtained and compared.

Scale-up of laboratory-scale CLC reactors is another important task necessary to develop an understanding of the potential and efficiencies of CLC. In this study, scaling laws were used as a guide to design and then build two different-sized CLC reactors. Testing of the reactors involved a focus on chemical similarities. Comparisons of the performance of both reactors showed good consistency, thereby validating the scale modeling method and the scale laws for CLC reactors.

KEYWORDS: Oxygen Carrier, Reactivity; Diffusion Mechanism; Solid Solution,  
Scale-up

Fang Liu

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Student's Signature

December 18, 2013

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Date

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## 1. Introduction

### 1.1 Background

In recent years, as the issues associated with atmospheric chemistry and the potential of global warming has come into focus, professional, industrial, governmental and private organizations and panels have conducted research to examine and expound future global weather scenarios that are based on well-founded theories, experiments and data. To date, a general consensus has been formed on how climate is changing and may further change into the future. It is known that the average earth surface temperature and sea levels are increasing while snow cover is decreasing; it is projected that the average temperature will continue to rise at a rapid rate <sup>1</sup>.

Greenhouse gases (GHG), such as water vapor (H<sub>2</sub>O), carbon dioxide (CO<sub>2</sub>), methane (CH<sub>4</sub>) and ozone (O<sub>3</sub>), have been established as main atmospheric constituents that can drive global warming, with CO<sub>2</sub> the most prevalent, and the most affected by human activity and industry <sup>2</sup>. Studies show that atmospheric CO<sub>2</sub> concentrations reached 400 ppm in 2013 <sup>3-5</sup>, or 40 % above pre-industrial levels. CO<sub>2</sub> emitted into the atmosphere by fossil fuel combustion is the most significant greenhouse gas contributing to climate change, and the use of coal alone accounted for 43 % of global CO<sub>2</sub> emission in 2010 <sup>6</sup>. The influence of CO<sub>2</sub> from industrial activity on increasing global CO<sub>2</sub> concentrations is widely accepted and points to the need to reduce emissions of this greenhouse gas. However, the number of options that could be employed to mitigate increases in CO<sub>2</sub> atmospheric levels is limited and will need to include both CO<sub>2</sub> capture and its storage.

## 1.2 CO<sub>2</sub> capture and disposal

Currently, three main technologies for CO<sub>2</sub> capture from power production have been examined: post-combustion, pre-combustion, and oxy-fuel combustion. In post-combustion technologies the CO<sub>2</sub> is captured from the flue gas after combustion of the fuel, and usually involves a two-step process that employs separation and regeneration technologies. During CO<sub>2</sub> separation from the flue gas, absorption, adsorption, and membrane or cryogenic processes could be employed <sup>7</sup>. Then, regeneration of the separation medium is used to release the captured CO<sub>2</sub> and to recycle the capture medium for re-use in CO<sub>2</sub> separation. However, typical CO<sub>2</sub> concentrations in flue gas for post-combustion processes are lower than 15 %; hence, the thermodynamic driving force for CO<sub>2</sub> capture is low and the costs are high <sup>8</sup>. As an example, the energy required for CO<sub>2</sub> capture using monoethanolamine (MEA) is estimated to reduce a pulverized coal combustion plant's output by about 30 % , which would equate to a very substantial 80% increase in the cost of electricity <sup>9</sup>.

Pre-combustion processing is usually applied to integrated gasification combined cycle (IGCC) using gas-liquid absorption <sup>10</sup>. The first step is to partially oxidize a fuel containing carbon and hydrogen to H<sub>2</sub> and CO. Steam is then injected into the reaction zone to induce the water-gas-shift reaction:  $\text{CO} + \text{H}_2\text{O} \rightarrow \text{CO}_2 + \text{H}_2$ , converting CO into CO<sub>2</sub> and generating high concentrations of H<sub>2</sub> which is a fuel having low greenhouse gas emission potential. The CO<sub>2</sub> is then separated using methods very similar to those used in post-combustion processing. However, CO<sub>2</sub> capture by pre-combustion processing is not



technically easy or inexpensive. For example, the energy penalty for pre-combustion CO<sub>2</sub> capture was calculated to be 18 %-19 %<sup>11</sup>.

Oxy-fuel combustion uses pure oxygen to combust fuels, and produces flue gas containing primarily CO<sub>2</sub> and water vapor. Water vapor is very easy to condense, resulting in a pure CO<sub>2</sub> stream which is ready to capture. However, an air separation unit (ASU) is required in oxy-fuel combustion to provide pure oxygen separated from air, and is a high-energy consuming process<sup>12</sup>. An alternative is to use a solid metal oxide as an oxygen carrier (OC). The method that adopts metal oxides to provide oxygen for oxy-fuel combustion is known as chemical looping combustion (CLC).

For all technologies to be used for CO<sub>2</sub> capture, it is necessary to compress the CO<sub>2</sub> under high pressure for sequestration or further use. The CO<sub>2</sub> can be used as a raw material in food production, firefighting, fish farming, and rubber and plastic processing, etc<sup>2</sup>. It can also be used to promote crude oil and natural gas production by injection into wells, helping to drive the crude oil and natural gas to surface processing equipment<sup>13</sup>. But the amount of CO<sub>2</sub> used by these types of industrial activity is very small relative to the vast amount generated during power production from hydrocarbon fuels<sup>14</sup>. It would therefore be necessary, if CO<sub>2</sub> capture technologies are employed for mitigating greenhouse gas emissions, to transport huge amounts and then to store it in secure and isolated underground locations for 100's-to-1,000's of years. Alternative to storage of CO<sub>2</sub> in geological formations, it may also be possible to store it through surface mineral carbonation or underwater. However, in all possible scenarios, the location and approach

must be thoroughly investigated before implementation because CO<sub>2</sub> leakage can seriously hurt people, damage property, make seawater more acidic and threaten sea animals and plant life.

Therefore, widespread, highly efficient and safe CO<sub>2</sub> emission abatement and storage technologies would be required for future control of atmospheric CO<sub>2</sub> concentration or mitigation of its emission during industrial processing. The task is daunting. Nevertheless, this dissertation examines a small but important segment of a promising approach for CO<sub>2</sub> capture from flue gas – chemical looping combustion (CLC).

### 1.3 Chemical looping combustion introduction

Among CO<sub>2</sub> capture technologies, CLC has been proposed to be one offering higher efficiencies and lower costs<sup>15-16</sup>. It is a type of oxy-fuel combustion having the advantage of *in situ* oxygen separability and low NO<sub>x</sub> emissions. In other words, CO<sub>2</sub> can be separated from flue gas inherently without the use of an energy-intensive, external ASU that is required for conventional oxy-fuel technology. Accordingly, CLC technology, as depicted in Figure 1, may be a promising method for fossil-based power generation in a carbon-constrained world.

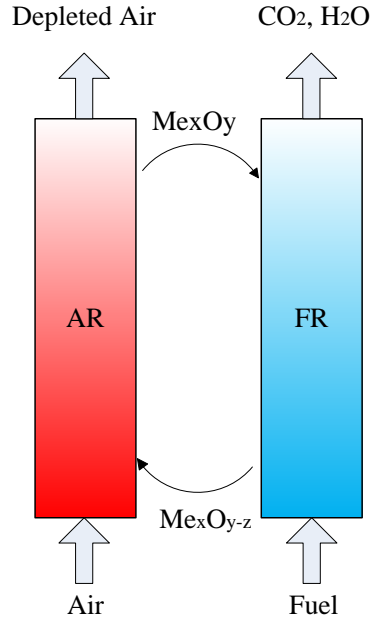
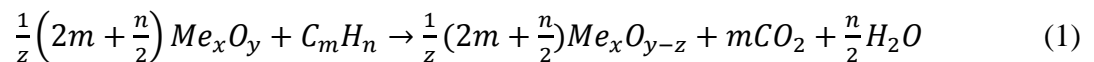


Figure 1 Ideal CLC diagram.

CLC uses a solid oxygen carrier (OC), usually a metal oxide, to provide oxygen for combusting fuel without the presence of nitrogen from air. The reduced OC is then recycled to an air reactor (AR) to be re-oxidized, and then it is re-used in the fuel reactor (FR) to provide oxygen in subsequent fuel combustion cycles. Through the use of the OC, the flue gas is separated into two parts. The air reactor gas outlet contains a high-temperature, oxygen-depleted gas containing mostly  $N_2$ . The fuel reactor outlet gas is primarily water vapor and  $CO_2$ . Because water vapor can be condensed very easily, an exhaust gas can be made that is highly concentrated in  $CO_2$  and readied for compression and storage.

The reactions in the fuel reactor and the air reactor can be expressed as:

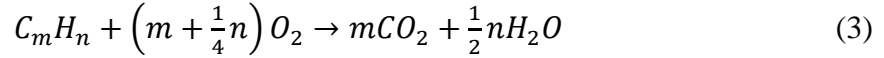
Fuel reactor:



Air reactor:



The overall reaction is:



#### 1.4 CLC combined power generation

CLC technology is suited for power generation because the flue gas from the air reactor is at high temperatures as is the flue gas from the fuel reactor<sup>17</sup>. Hence, gas and CO<sub>2</sub> turbines can be used to generate power.

Recent studies have shown that CLC with CO<sub>2</sub> capture would have higher efficiency than traditional pulverized coal (PC) power generation under similar conditions<sup>18-23</sup>. According to a Department of Energy (D.O.E) report<sup>24</sup>, a CLC-based CO<sub>2</sub> capture system could give higher overall efficiency than that of post-combustion CO<sub>2</sub> capture or alternative oxy-fuel technologies. A pressurized CLC, super-critical power plant would have the highest efficiency up to 46 %, which is almost double that achieved in sub-critical PC plants equipped with an amine scrubber (~25 % efficiency) or of an oxy-fuel PC system (~27 %), and is about 50 % greater than that of a state-of-the-art IGCC providing CO<sub>2</sub> enrichment (36 % efficiency). Naqvi et. al.<sup>19</sup> designed a detailed CLC power generation plant using natural gas as fuel, and analyzed the thermodynamic cycle of this plant based on a steady state model. Their results show that a CLC power generation plant could achieve an overall efficiency of 49.7 % with zero CO<sub>2</sub> emissions. With such advantages, CLC has attracted intensive attention in the last decade, with bench scale and pilot scale testing successfully showing the feasibility of CLC processing

using various OC's and fuels. However, CLC technology still has a long way to go before being applied commercially. One of the main bottlenecks is the OC itself because excellent reactivity and stability are required for overall economic viability of CLC processing<sup>2, 17</sup>. Furthermore, appropriate methods for reactor scale-up also need to be figured out.

## 2. Literature Review

### 2.1 Oxygen carrier introduction

Since the beginning of CLC research, OC development has always been a key focus of study due to the concern of slow reaction rates in the fuel reactor. Active metal oxides would rapidly provide oxygen during the reduction reaction, and re-obtain oxygen during oxidation in the air reactor, hence the name oxygen carrier (OC). The transition metals, such as Ni, Co, Cu, and Fe, have several different valence states; they are usually used as OCs. The performance of OCs is a key point for successful CLC processing, the requirements of which can be described as:

- (1). High oxygen transfer capacity and high reaction rate;
- (2). High stability/durability;
- (3). Good mechanical strength and low attrition rate;
- (4). Cost-effective and environmentally acceptable.

The OCs can be produced using several different methods, with freeze granulation, impregnation and spray-drying the most popular<sup>8, 25-31</sup>. The OCs produced by these methods are usually called synthetic OCs in contrast to natural OCs from, for example, ilmenite ( $\text{FeTiO}_3$ ) which can be used after crushing mined ores; typically, natural OCs are less expensive than synthetic OCs. For synthetic OCs, inert materials like  $\text{Al}_2\text{O}_3$ ,  $\text{SiO}_2$ ,  $\text{TiO}_2$ , YSZ and  $\text{ZrO}_2$  are usually used as supports for the active transition metal oxides. The support works as a binder to increase the mechanical strength and helps to disperse the metal oxide which enables increased surface areas. Hence, the reactivity and durability of OCs can be enhanced by using support materials.

## 2.2 Performance evaluation

The performance of OCs is evaluated using the following parameters and methods.

### (1) Oxygen transport capacity, $R_o$

The oxygen transport capacity,  $R_o$ , is an important property of OCs, and is defined as the mass fraction of usable oxygen in the OCs between the air reactor and fuel reactor:

$$R_o = (m_{ox} - m_{red})/m_{ox} \times 100 \% \quad (4)$$

where  $m_{ox}$  is the mass of the OC in an oxidized state in grams and  $m_{red}$  is the mass of the OC in its reduced state in grams.

The oxygen transport capacity of several metal oxides is listed in Figure 2. The CuO, NiO and CoO have relative higher oxygen transfer capacities; however, Fe<sub>2</sub>O<sub>3</sub> has the highest theoretical oxygen transfer capacity. If Fe<sub>2</sub>O<sub>3</sub> was reduced to metallic iron (Fe), it would have an oxygen transport capacity of 30 %.

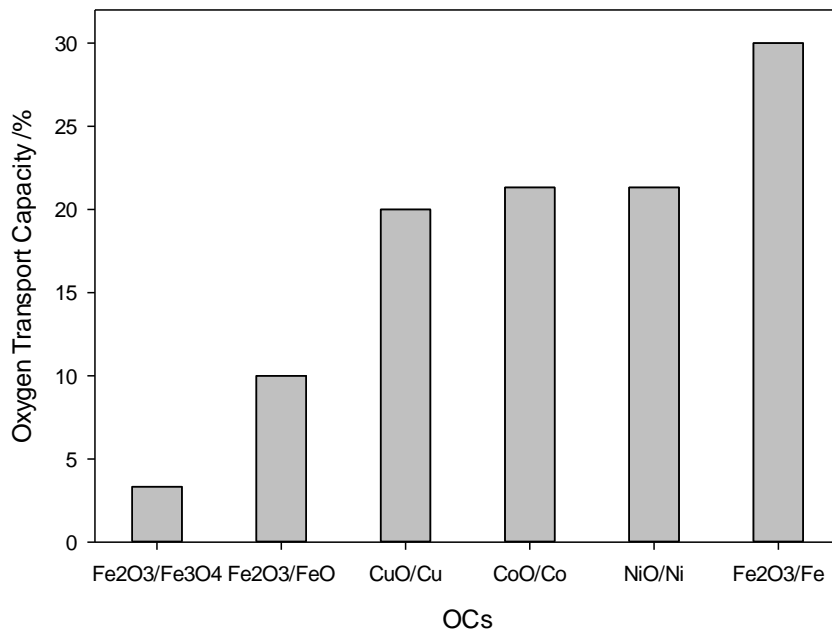


Figure 2 Theoretical oxygen transfer capacity.

(2) Conversion of reduction,  $X_r$ , and reaction rate,  $w$

Conversion of reduction is the degree of OC reduction, and can be defined as the actual mass loss of OC divided by the mass of oxygen that the OC could provide:

$$X_r = (m_{ox} - m)/(m_{ox} - m_{red}) \times 100 \% \quad (5)$$

The reaction rate  $w$  is defined by time rate of change of conversion:

$$w = dX_r/dt \quad (6)$$

(3) Mechanical strength

Mechanical strength reflects, to some degree, the particles resistance to attrition stresses. Particles with higher mechanical strength possess a higher ability to retain their initial shape, and thus would exhibit better durability. It is worth noting that, for the OC mechanical strength to be comparable in a study, a narrow particle size distribution must be used.

### 2.3 OC screening

In the last two decades, a number of institutes have begun working in OC development research; Table 1 lists the major affiliations of these institutes and their OC product. Most of the OCs are synthetic, with NiO, CuO or Fe<sub>2</sub>O<sub>3</sub> as the active metal oxides. Natural OCs such as ilmenite (FeTiO<sub>3</sub>) have also been studied. The OCs were investigated in TGA or bench scale reactors using varies fuels.

Fan<sup>32</sup> compared the performance of NiO, CuO and Fe<sub>2</sub>O<sub>3</sub> based OCs, as shown in Table 2. Nickel based OCs have the advantage of fast reaction rates and a fair melting point, however, nickel oxides lack mechanical strength and are toxic to humans and the environment. Copper oxides also show high reactivity, but tend to agglomerate<sup>27</sup>. Iron



based OCs have attracted most research attention in recent years, mainly due to their inexpensive and non-toxic nature, they also have sufficient reaction rates. The advantages of low cost and high mechanical strength of iron based OCs led to their use in a commercial CLC project<sup>2, 26, 33</sup>. Our recent study showed that, in comparison to the other two metal oxides (NiO and CuO), Fe<sub>2</sub>O<sub>3</sub> based OCs exhibited much better resistance to water vapor degradation<sup>34</sup>. Hence, Fe<sub>2</sub>O<sub>3</sub> is currently believed to be the most suitable material for OC development.

#### 2.4 Thermodynamic analysis

The reaction in the fuel reactor is reduction, which can be endothermic or exothermic. The reaction in the air reactor is oxidation, usually exothermic. With Fe<sub>2</sub>O<sub>3</sub>, the reactions in a CLC using gaseous fuels such as CO and oxidants (O<sub>2</sub>) are listed in Table 3. Fe<sub>2</sub>O<sub>3</sub> can be reduced to Fe<sub>3</sub>O<sub>4</sub>, FeO and even the metallic iron (Fe) depending on the degree of reduction<sup>35</sup>.

The heat of reaction (in this case, heat of combustion) can be calculated from:<sup>36</sup>

$$\Delta H(T^0) = \sum_i (v_i'' - v_i') H_i^o(T^0) \quad (7)$$

Where:

$\Delta H$  is heat of reaction, kJ mol<sup>-1</sup>;

$v_i''$  is mole concentration coefficient of product;

$v_i'$  is mole concentration coefficient of reactant;

$H_i^o$  is heat of formation, kJ mol<sup>-1</sup>.

The heat of combustion for related reactions is also listed in Table 3. Most of these reactions are exothermal, which is beneficial when designing the reactor because less external-supplied thermal energy would be needed in the fuel reactor.

Table 1 CLC research institutes and the OCs

Institute	Researchers	OCs (active metals)	Fuel
Tokyo Institute of Technology	Nakano, Ishida, Jin, Okamoto	NiO, Fe <sub>2</sub> O <sub>3</sub>	H <sub>2</sub> , CO, CH <sub>4</sub>
Chalmers University of Technology	Lyngfelt, Zafar Mattisson, Johansson, Leion,	NiO, Fe <sub>2</sub> O <sub>3</sub> , CuO, Ilmenite, Hematite, Sandvik,	natural gas, CH <sub>4</sub> , syngas, petroleum coke, charcoal,
Institute de Carboquimica	Abad, de Diego, Beatriz	NiO, Fe <sub>2</sub> O <sub>3</sub> , CuO, Ilmenite,	natural gas, CH <sub>4</sub> , syngas
Ohio State University	Fan	Fe <sub>2</sub> O <sub>3</sub>	solid fuel
Korea Institute of Technology	Son	Fe <sub>2</sub> O <sub>3</sub>	CH <sub>4</sub>
Southeast University	Xiao, Gu	CaSO <sub>4</sub> , Fe <sub>2</sub> O <sub>3</sub>	syngas, coal
Technical University of Viena	Kolbitsch	NiO, CoO	CH <sub>4</sub> , syngas
Alstom	Andrus	CaSO <sub>4</sub>	Coal
University of Kentucky	Kunlei Liu	Fe <sub>2</sub> O <sub>3</sub> , Ilmenite	CO, syngas, coal char

Table 2 Comparison of Ni, Cu and Fe- based OCs

OC	Ni- based	Cu- based	Fe- based
Reactivity	good	good	low
Melting point	fair	low	high
Strength	low	fair	good
Toxic	yes	no	no
Cost	high	high	low

Table 3 Reactions and heat of combustion

Reaction	Heat of Combustion
	$\Delta H$ (kJ/mol fuel)
$3\text{Fe}_2\text{O}_3 + \text{CO} = 2\text{Fe}_3\text{O}_4 + \text{CO}_2$	-48.3
$\text{Fe}_2\text{O}_3 + \text{CO} = 2\text{FeO} + \text{CO}_2$	-1.5
$\text{Fe}_2\text{O}_3 + 3\text{CO} = 2\text{Fe} + 3\text{CO}_2$	7.8
$4\text{Fe}_3\text{O}_4 + \text{O}_2 = 6\text{Fe}_2\text{O}_3$	-117.4
$4\text{FeO} + \text{O}_2 = 2\text{Fe}_2\text{O}_3$	-140.8
$4\text{Fe} + 3\text{O}_2 = 2\text{Fe}_2\text{O}_3$	-412.3

\*Heat of formation can be found in NIST-JANAF Thermochemical Tables.<sup>37</sup>

## 2.5 Iron oxide based OC review

Natural iron oxide OCs such as hematite have low cost and are easily mined and processed; their reactivities have been studied by several researchers<sup>25-26, 38-41</sup>. The results showed they were suitable as OCs in CLC, but the ores had a tendency to agglomerate at high temperature<sup>2</sup>, which is problematic for CLC applications. Another possibly important natural OC ore is the ilmenite. It is cheap, abundant and shows good reactivity and oxygen transport capacity<sup>42-47</sup>. It is also recommended to pre-calcine the fresh ilmenite which increases its porosity and mechanical strength. However, a possible disadvantage of it is that defluidization (agglomeration) may occur when the particles are in a highly reduced state<sup>48-49</sup>.

Synthetic OCs like Fe<sub>2</sub>O<sub>3</sub> supported on Al<sub>2</sub>O<sub>3</sub> has been studied in some research. They show good reactivity and mechanical strength, and are better than Fe<sub>2</sub>O<sub>3</sub> on TiO<sub>2</sub> and kaolin<sup>8, 30, 50</sup>. Although Fe<sub>2</sub>O<sub>3</sub> on Al<sub>2</sub>O<sub>3</sub> shows some good properties, it also has some disadvantages. The first one is its tendency to agglomerate<sup>27, 50</sup>. Another disadvantage is that Fe<sub>2</sub>O<sub>3</sub> will react with the Al<sub>2</sub>O<sub>3</sub> support and form FeAlO<sub>4</sub><sup>2</sup>; this molecular form has decreased oxygen transfer ability.

To avoid the formation of iron aluminate, MgAl<sub>2</sub>O<sub>4</sub> has also been investigated as a support material; it is a beneficial support because the MgO in the Al<sub>2</sub>O<sub>3</sub> eliminates interactions between the iron oxide and the support and helps to maintain the iron oxides' reactivity and durability. Johansson et al<sup>51</sup> studied Fe<sub>2</sub>O<sub>3</sub> on MgAl<sub>2</sub>O<sub>4</sub> in a fluidized bed and examined the influence on OC properties. Their preparation included freeze granulation methods and they used a reductive gas which was 50 vol. % CH<sub>4</sub> balanced

with H<sub>2</sub>O. The results showed an optimum OC composition was 60 wt. % Fe<sub>2</sub>O<sub>3</sub> and 40 wt. % MgAl<sub>2</sub>O<sub>4</sub>, and when sintered at 1100 °C; it had high reactivity and was not easily agglomerated or attritted. Leion et al <sup>49, 52</sup> continued to test this OC using petroleum coke as fuel in a small lab reactor, and showed that the OC reacted quickly with intermediate gasification products such as CO and H<sub>2</sub>; the addition of the OC greatly enhanced coke gasification. Zafar et al <sup>29</sup> compared the redox reactivity of Ni, Cu, Fe and Mn supported on SiO<sub>2</sub> and MgAl<sub>2</sub>O<sub>4</sub>, and concluded that Fe on SiO<sub>2</sub> was not a feasible OC because of silicate formation, whereas Fe<sub>2</sub>O<sub>3</sub> supported on MgAl<sub>2</sub>O<sub>4</sub> showed high reactivity.

It has been shown that bi-metal oxide OCs usually work better than mono metal oxide OCs <sup>53</sup>. For example, Johansson et al <sup>53</sup> studied synergistic effects of mixed oxides of nickel oxide, 60 wt. % NiO and 40 wt. % MgAl<sub>2</sub>O<sub>4</sub> sintered at 1400 °C (labeled as N6AM1400), and iron oxide, 60 wt. % Fe<sub>2</sub>O<sub>3</sub> and 40 wt. % MgAl<sub>2</sub>O<sub>4</sub> sintered at 1100 °C (F6AM1100). They used 3 wt. % nickel oxide mixed with 97 wt. % iron oxide and conducted tests using a laboratory-scale fluidized bed of quartz with methane as the fuel. The results showed that the CO<sub>2</sub> production rate was twice as much as the sum of the CO<sub>2</sub> production rates when the two oxygen carriers were tested separately. Rydén, et al. <sup>54</sup> did similar experiments using a batch fluidized bed reactor and a circulating fluidized bed reactor, with additives of N6AM1400 or N18- $\alpha$ -Al<sub>2</sub>O<sub>3</sub> (18 wt. % NiO supported on Al<sub>2</sub>O<sub>3</sub>) to ilmenite with different concentrations; the results were that a small amount of nickel oxide added to natural ilmenite increased the overall activity greatly.

Compared to NiO, CuO- based OCs, Fe<sub>2</sub>O<sub>3</sub>-based OCs are known to have relatively lower reactivity <sup>27, 55</sup>. However, iron oxide is more abundant and less costly, and is non-

toxic <sup>56</sup>, and could become more advantageous than these other oxides for commercial scale use if its reactivity could be enhanced. OCs with bi-metal oxide compositions have been shown to be more effective than mono-metal oxides. For example, recent studies have shown the attributes of mixed metal oxide OCs, including: CoO and NiO supported on Al<sub>2</sub>O<sub>3</sub> <sup>57</sup>; CuO and NiO supported on Al<sub>2</sub>O<sub>3</sub> <sup>58</sup>; Fe<sub>2</sub>O<sub>3</sub> and NiO supported on Al<sub>2</sub>O<sub>3</sub> or MgAl<sub>2</sub>O<sub>4</sub> <sup>30, 53</sup>; Fe<sub>2</sub>O<sub>3</sub> and MnO<sub>2</sub> without support or supported on ZrO<sub>2</sub> <sup>59-61</sup>; and Fe<sub>2</sub>O<sub>3</sub> and CuO supported on Al<sub>2</sub>O<sub>3</sub> <sup>62-64</sup>. Compared with these mixed metal oxides, CeO<sub>2</sub> as an additive to OCs has not been extensively studied even though it may impart unique characteristics to iron oxide based OCs.

CeO<sub>2</sub> has a fluorite structure, with each Ce<sup>4+</sup> surrounded by eight equivalent, nearest O<sup>2-</sup> ions that form the corners of a cube and coordinated to four Ce<sup>4+</sup> ions <sup>65</sup>. When the Ce<sup>4+</sup> ions are replaced by lower valence cations, oxygen vacancies will be created <sup>66</sup>. If two cerium ions are replaced by trivalent ions - for example, by two trivalent Ce<sup>3+</sup>, an oxygen vacancy or lattice defect is created that can be the most reactive site on the surface of the metal oxide <sup>67</sup>. Both surface and bulk oxygen vacancies occur in CeO<sub>2</sub> and are suitable sites for adsorption <sup>68</sup>. Hence, besides being useful as a support for catalytic surfaces, CeO<sub>2</sub> also can actively participate in chemical reactions.

CeO<sub>2</sub>-containing OCs have shown promising results in CLC testing <sup>69-72</sup>, in CH<sub>4</sub> reforming for H<sub>2</sub> and CO production <sup>73-77</sup> (chemical looping reforming), and in CO<sub>2</sub> splitting <sup>78-79</sup>. Miller et al <sup>72</sup> added 5 wt. % CeO<sub>2</sub> to hematite (Fe<sub>2</sub>O<sub>3</sub>) and found improved OC performance for CH<sub>4</sub> oxidation; the CeO<sub>2</sub> did not affect the surface area of the hematite. Li et al <sup>76</sup> used Fe<sub>2</sub>O<sub>3</sub> and CeO<sub>2</sub> materials (Fe : Ce molar ratio =3:7) during the conversion of CH<sub>4</sub> to synthesis gas and reported that Fe<sub>2</sub>O<sub>3</sub> and CeO<sub>2</sub> formed a solid

solution having higher activity and selectivity than hematite by itself. Galvita et al <sup>79</sup> investigated the effect of CeO<sub>2</sub> upon Fe<sub>2</sub>O<sub>3</sub> during H<sub>2</sub>-CO<sub>2</sub> redox reactions for CO<sub>2</sub> utilization and found enhanced reaction capacity and increased stability relative to Fe<sub>2</sub>O<sub>3</sub> by itself. However, the mechanism by which CeO<sub>2</sub> promotes Fe<sub>2</sub>O<sub>3</sub>-based OCs during CLC is still not clear. Miller et al <sup>72</sup> theorized that CH<sub>4</sub> first reacted with CeO<sub>2</sub> lattice oxygen to form CO and H<sub>2</sub>, and then the reduced cerium oxide promoted CH<sub>4</sub> decomposition to active intermediates such as C and H<sub>2</sub> which, in turn, facilitated a greater use of oxygen from the natural ore hematite. Galvita et al <sup>79</sup>, on the other hand, attributed the promotional role of CeO<sub>2</sub> to that of suppressing the sintering of iron oxide. Hence, significant differences currently exist on the promotional role of CeO<sub>2</sub>.

No standard exists to evaluate the performance of OCs. Liu et al <sup>35</sup> recommended the use of FG Fe<sub>2</sub>O<sub>3</sub> 50 OC (freeze granulated OC with the composition of 50 wt. % Fe<sub>2</sub>O<sub>3</sub> and 50 wt. % Al<sub>2</sub>O<sub>3</sub>) as a reference because it has been very widely studied and has a moderate Fe<sub>2</sub>O<sub>3</sub> concentration. It has demonstrated a reasonable reactivity in the presence of water vapor and CO<sub>2</sub>, was not highly deactivated due to coal impurities, has high mechanical resistance to attrition, and is cost effective for production. This OC is used as a reference in this dissertation.

## 2.6 Scale modeling

Reactor scale-up is a major task for engineers and is a fundamental step in the realization and optimization of industrial plants <sup>80</sup>. Scale-up is able to transfer information from equipment of one size to other similar equipment having a different size <sup>81</sup>. Kuwana et al <sup>82</sup> gave an example of understanding phenomena happening in a real process by doing a scale modeling study of a laboratory setup. Scale modeling does not only implies the

capacity of designing and operating large plants but also the skill of developing new and more efficient reaction technologies that would be cost and product quality competitive and meet environmental requirements <sup>80</sup>.

CLC is a novel combustion technology that could capture CO<sub>2</sub> inherently with very little energy penalty. Research on scale modeling of CLC would significantly benefit the commercialization of this technology. However, scale-up of fluidized bed reactors is known to be more difficult than that of other types of reactors <sup>83</sup>. This difficulty is because of the complicated fluidization state inside the reactor and the many possible operating parameters. The history of fluidization scale-up has examples of success <sup>84-86</sup>; but severe failures have also been reported <sup>83,87</sup>. Even today, the scale-up of fluidized bed technology is challenging <sup>87</sup>. In other words, just as Matsen <sup>88</sup> stated, ‘scale-up is still not an exact science, but is rather a mix of physics, mathematics, witchcraft, history and common sense that we call engineering’.

Development and commercialization of any new chemical process will cost significant amount of time and effort, and requires substantial capital expenditures. Using proven techniques based on experience and mathematic and/or design models can minimize the risk and uncertainty when scaling up fluidized bed technology <sup>89</sup>. Currently, no open literature studies are known that have studied CLC reactor scale-up based on scaling laws.



## 2.7 Objectives

In this study, we will examine the development of novel OCs by using small amounts of  $\text{CeO}_2$  as an additive to a  $\text{Fe}_2\text{O}_3$  based OC. The objectives of this study are:

- (1) To understand the reactivity of iron oxide based OCs by using TGA-MS and a bench scale CLC experiment setup;
- (2) To investigate the effects of  $\text{CeO}_2$  additive on particles' physical properties (morphology, BET surface area, mechanical strength);
- (3) To elucidate and then discuss the mechanisms of the promoting role of  $\text{CeO}_2$  in  $\text{Fe}_2\text{O}_3$  based OC;
- (4) To obtain the kinetic parameters of OCs;
- (5) To perform a scale-up study of a CLC bench scale reactor, and to gain experience and insight on CLC fluidized bed reactors during scale-up, and to find potential correlations that would be beneficial for future CLC reactor scale-up.

### 3. Experimental

#### 3.1 Oxygen carrier

Commercially available metal oxides were used during OC preparation. Aluminum oxide powder (Sigma-Aldrich, standard grade), iron oxide (Sigma-Aldrich, 99.0 % purity) and cerium oxide (Strem Chemicals, 99.9 %) were used to prepare OC materials by freeze granulation methods. Typically, a slurry was prepared by ball milling mixtures of metal oxides, dispersant (A40) and binder (PVA) within de-ionized water. The well-mixed slurry was sprayed through a nozzle into liquid nitrogen, leading to the formation of frozen spherical particles which were then dried in a freeze dryer (Virtis Advantage Plus). After freeze drying, the particles were calcined in air at 1400 °C for 6 hours. Three types of OC, labeled as OC #1, OC #2 and OC #3, were produced for this study, as shown in Table 4. The OCs were selected with a diameter between 150-300  $\mu\text{m}$ .

#### 3.2 Microscopic imaging

The morphology of the OC particles was examined using a Hitachi S-4800 scanning electron microscope. The OC particles were immobilized on the sample stage with conductive double sided carbon tape. The accelerating voltage was set to 15 kV.

#### 3.3 Mechanical strength

The mechanical strength was measured in a Shimpo FGE-10X instrument which can record the peak force every time a particle was crushed. The mechanical strength was obtained by averaging 30 measurements using particles that had been randomly extracted from samples and having a particle size distribution range of the reaction tested samples (150-300  $\mu\text{m}$ ).

Table 4 Composition and sintering temperature of OCs

OC	Composition	Sintering Temperature
#1	10 wt. % CeO <sub>2</sub> , 40 wt. % Al <sub>2</sub> O <sub>3</sub> , 50 wt. % Fe <sub>2</sub> O <sub>3</sub>	1400°C
#2	50 wt. % Fe <sub>2</sub> O <sub>3</sub> , 50 wt. % Al <sub>2</sub> O <sub>3</sub>	1400°C
#3	10 wt. % CeO <sub>2</sub> , 90 wt. % Fe <sub>2</sub> O <sub>3</sub>	1400°C

### 3.4 Surface area measurement

A Micromeritics ASAP 2020 surface analyzer was employed to examine microstructure of the OC samples. The surface area is based on the Brunauer-Emmett-Teller (BET) method. About 450 mg sample was used for each measurement. The sample was degassed at 160 °C for overnight, and then it was subjected to the isothermal measurement by N<sub>2</sub> adsorption-desorption at 77 K.

### 3.5 XRD

In order to investigate the crystalline phase of the samples, the X-ray diffraction patterns were obtained using a Rigaku SmartLab system in the 2 theta range of 20-90° with CuKα irradiation.

### 3.6 Raman spectroscopy

Raman spectroscopy (Horiba-Jobin Yvon LabRam HR) was used to investigate the molecular speciation of the OCs in their oxidized and reduced forms. Spectral resolution was approximately 2 cm<sup>-1</sup>; wavenumber calibration was checked using the silica vibrational mode at 520.7 cm<sup>-1</sup>.

### 3.7 TGA/MS measurement

The CLC reaction testing included the use of a TGA-MS system (Netzsch STA449C and Netzsch QMS 403C) and a bench-scale CLC setup. The TGA-MS system is shown in Figure 3; in it was placed about 500 mg ( $\pm 1$  mg) of OC and then the OC was alternately oxidized or reduced using O<sub>2</sub> or CO diluted in argon, respectively. Mass flow controllers (MFCs) were used to control the flow rate of the feed gases and a LabView program was designed and used to control and monitor the MFCs. The redox cycles followed a sequence of: (i) 5 min argon purge at 200 ml min<sup>-1</sup>; (ii) 20 min oxidation with 20 vol. % O<sub>2</sub> balanced with argon, at 200 ml min<sup>-1</sup>; (iii) 5 min argon purge at 200 ml min<sup>-1</sup>; and (iv) 30 min reduction with 20 vol. % CO balanced with argon, at 200 ml min<sup>-1</sup>. The exhaust gas from the TGA was analyzed by mass spectrometry (MS) operated in a MID (multiple ion detection) mode.

### 3.8 Bench scale CLC setup

This bench scale test platform can provide not only important information for the performance of the OCs under realistic conditions as exist in commercial scale units, but also provide the experience for further scale-up and troubleshooting. Its diagram is presented in Figure 4. It was comprised of a single fixed bed reactor of 5 cm inner diameter, and was used to simulate the oxidation and reduction of a CLC system by use of a switching valve that alternately enabled either oxidizing or reducing gas into the bottom of the reactor. During each test about 100 g of the OCs were placed onto a height-adjustable distributor at the bottom of reactor; the bed height of the OCs was approximately 7.5 cm. A K-type thermocouple was immersed into the center of the OC bed. Based on preliminary testing, it was determined that a 5 l min<sup>-1</sup> flow rate of feed gas

was appropriate to obtain a desired fluidization state. A Rosemond X-stream infrared multi-channel gas analyzer was used to measure the composition ( $O_2$ , CO,  $CO_2$  and  $CH_4$ ) of off-gas from the reactor. The experimental equipment used during reactor testing was controlled and data collected (temperature and gas concentrations) via a specially-designed LabView program. The furnace temperature can be set up to 1200 °C.

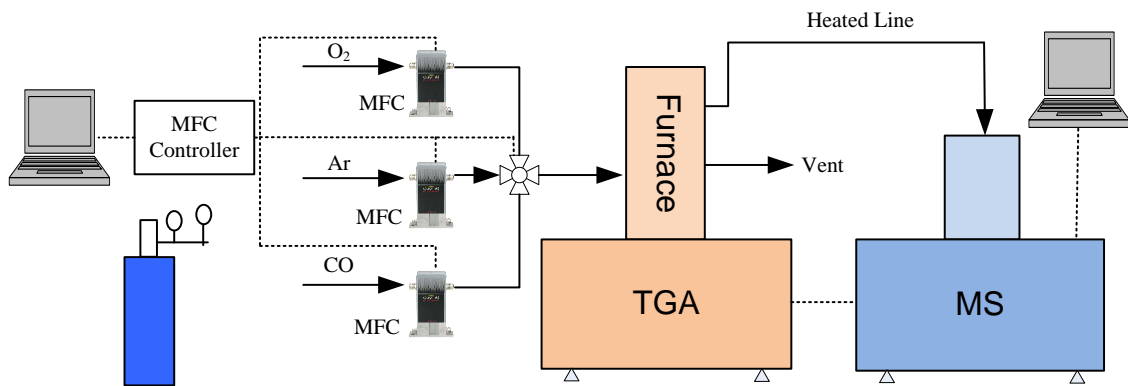


Figure 3 TGA-MS system.

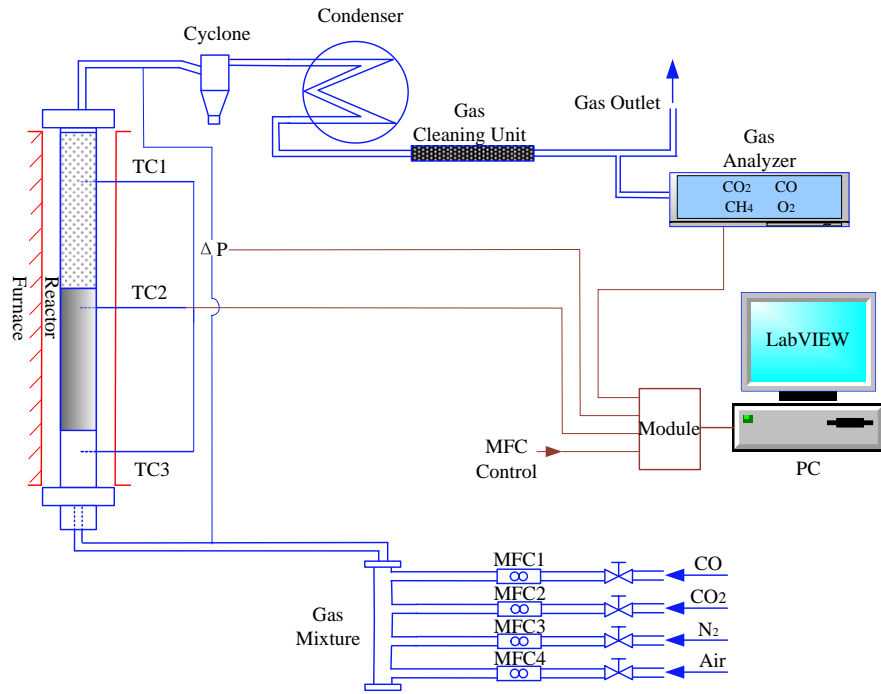


Figure 4 Bench scale CLC setup.

## 4. Reactivity Study

### 4.1 Effect of CeO<sub>2</sub> additive on the physical properties

The effect of CeO<sub>2</sub> on the physical properties of Fe<sub>2</sub>O<sub>3</sub>-based OCs was investigated first. Specifically, OC #1 and OC #2 particles were characterized by SEM, BET surface area and mechanical strength data acquisition. The SEM images of OC #1 and OC #2, presented in Figure 5, show that the OCs had very similar morphologies. The particles are spherical with a homogeneous, smoothed grain structure.

The BET surface area data of both OCs are shown in Figure 6. They had very small surface areas (0.34 m<sup>2</sup> g<sup>-1</sup> for freshly-calcined OC #1 and 0.31 m<sup>2</sup> g<sup>-1</sup> for freshly-calcined OC #2) typical of that caused by high temperature sintering. Upon being subjected to 5, 10 and 15 redox cycles, the surface areas were increased to 1.6 m<sup>2</sup> g<sup>-1</sup> but insignificant differences were found between the values for OC #1 and OC #2. Based on these data it was concluded that the addition of CeO<sub>2</sub> did not appreciably affect morphologies or surface areas of the OCs.

Structural properties can be further examined by carrying out mechanical strength measurements. Mechanical strength data of freshly-calcined OCs and after 5, 10 and 15 redox cycles are shown in Figure 7. Their mechanical strengths were identical, within the measurement uncertainties. Furthermore, mechanical strength did not change upon redox cycling. Hence, it is concluded that the addition of CeO<sub>2</sub> did not affect the OCs' mechanical strength.

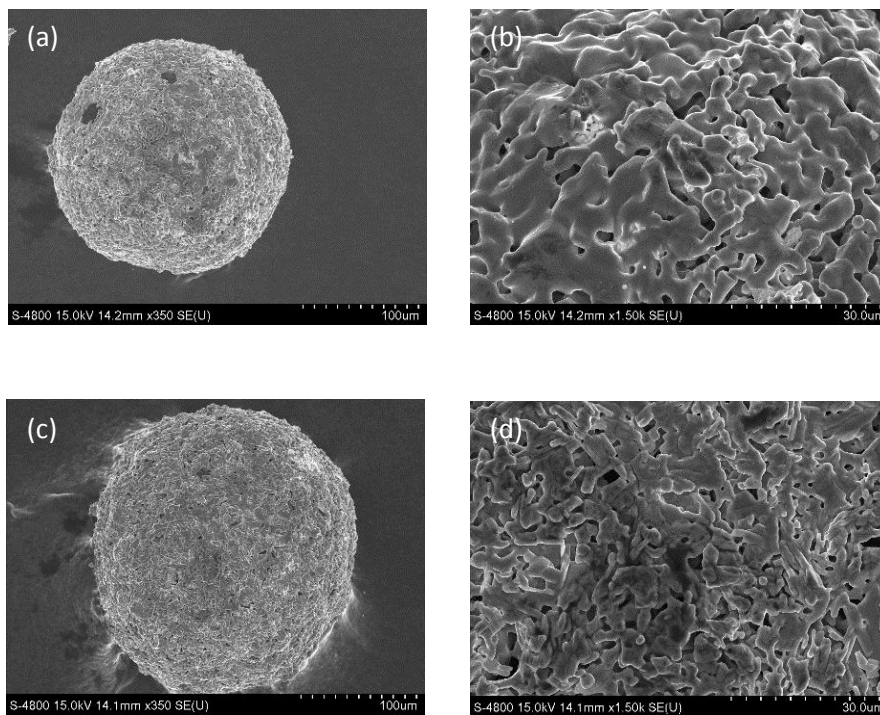


Figure 5 SEM images; (a) OC #1, 350; (b) OC #1, 1.5k; (c) OC #2, 350; (d) OC #2, 1.5k.  
The accelerating voltage was set to 15 kV, current was 20 µA.



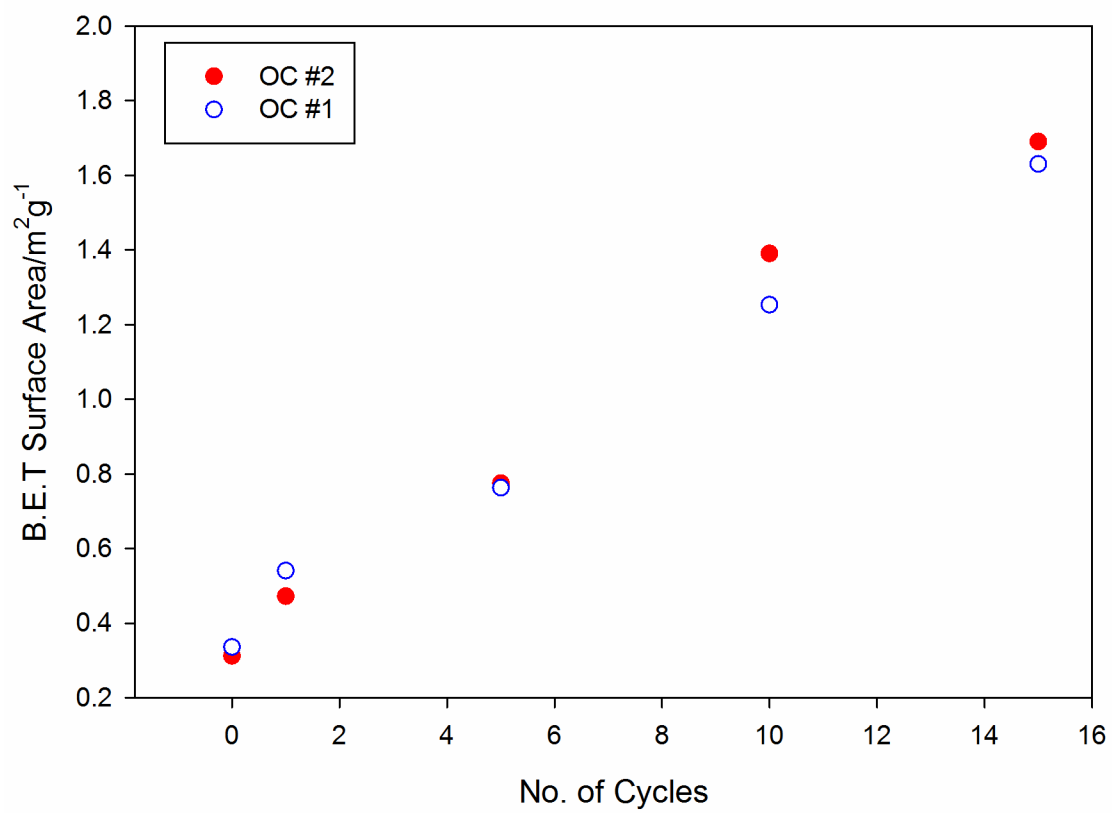


Figure 6 BET surface area.

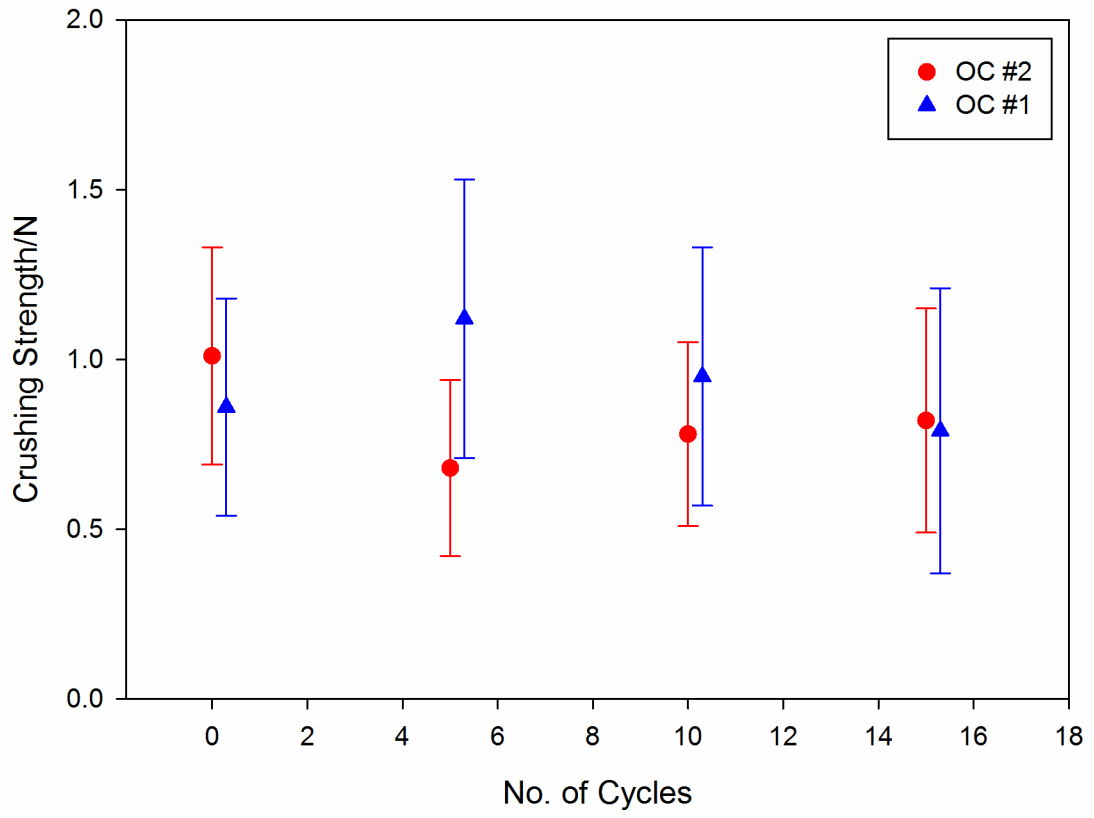


Figure 7 Mechanical strength.

## 4.2 Reactivity

The TGA-MS system and the bench-scale CLC reactor were used to investigate redox reactivity of the OC particles. Because oxidation reactivity is usually much faster than reduction reactivity, the reduction reactivity was the primary focus of this study. Before testing of the OCs, the TGA response baseline (without any samples) was measured to determine if fluctuations in apparent weights occurred during the switching between redox gas compositions. Also, although  $\text{Al}_2\text{O}_3$  is usually considered an inert support, its weight loss was tested under the redox gases to measure if any weight loss or addition occurred during gas composition changes. Because  $\text{CeO}_2$  can be partially reduced, it was necessary to prepare a reference sample of  $\text{CeO}_2$  and  $\text{Al}_2\text{O}_3$  and then to measure its weight loss or gain during redox cycling. The composition of this reference sample was 10 wt. %  $\text{CeO}_2$  and 90 wt. %  $\text{Al}_2\text{O}_3$ , and was prepared following the OC production procedures and conditions described in the foregoing description. During each test, the reactivities were evaluated by averaging reaction rates over a 30 minute reaction exposure.

Determination of the baseline data was identical to that used for the OCs, with four steps during each cycle: purge, oxidation, purge and reduction, using the same flow rates and gas compositions. These baseline test results are shown in Figure 8. Nine redox cycles were used during the TGA baseline testing, the results of which were very consistent. In general, these TGA data showed only minor perturbations when the gas composition was switched, and this perturbation was repeatable. In contrast, the OC baselines were conducted for 20 redox cycles, the data from which were also very consistent and repeatable. Although it was anticipated that  $\text{CeO}_2$  could lose oxygen during reduction at

high temperature, only a 0.1 % weight loss was measured for the OCs after the TGA baseline data were deducted. Hence, oxygen loss from  $\text{CeO}_2$  is projected to be very limited and  $\text{CeO}_2$  by itself could not be used as a main oxygen supplier during CLC.

The averaged reaction rate of OC #1 and OC #2 are shown in Figure 9 after the baseline was subtracted. Typical to these types of materials, several redox cycles were necessary to activate the OCs<sup>42, 90</sup>, as can be seen in the reaction rate data in which five redox cycles for OC #1 and nine cycles for OC #2 were necessary before a constant and repeatable reaction rate occurred during each cycle. From Figure 9 it is evident that the OC with 10 %  $\text{CeO}_2$  additive had better average reaction rate than did the other OC which was 7.7% higher than that of OC #2.

Figure 10 shows  $\text{O}_2$  and  $\text{CO}_2$  intensities from MS data during the 10<sup>th</sup> redox cycle for OC #1, OC #2 and baseline (no sample); higher current signals correspond to higher gas concentrations. During reduction (part III), both OC#1 and OC #2 produced about 10 times greater  $\text{CO}_2$  current intensity than the baseline testing. At the beginning of reduction, the  $\text{CO}_2$  concentrations from both OCs were very close; this response could be caused by a similar availability of surface oxygen on the OCs because both OCs had the same concentration of  $\text{Fe}_2\text{O}_3$ , and the same particle sizes and surface areas. Although after several minutes of reaction the  $\text{CO}_2$  concentrations from both OCs decreased below their concentrations at the beginning, the rate of change was different for both OCs such that the  $\text{CO}_2$  concentration from OC #1 was higher than from OC #2. This difference continued to be magnified throughout the reduction cycle, a trend which would not be expected if both OCs had equal reaction efficiencies or equal oxygen availabilities. If most of the surface oxygen was consumed during the first minutes into the reduction,

then only bulk oxygen would be sustaining CO<sub>2</sub> production, and it would have to diffuse to the surface from the bulk of the OC to oxidize the CO. Furthermore, a decrease of CO<sub>2</sub> concentrations from both OCs would be expected after minutes into the reduction because bulk diffusion of oxygen and then reaction with CO is slower than the consumption of surface oxygen for CO oxidation. Hence, the higher CO<sub>2</sub> concentration from OC #1 than from OC #2 suggests that OC #1 provided faster oxygen transfer than did OC #2. This conclusion is consistent with the TGA results that OC #1 lost more weight than did OC #2 during reduction, and with the fact that more time was needed for the O<sub>2</sub> signal to stabilize during oxidation of OC #1 (part I, Figure 10). The result also suggests that reduced OC #1 absorbed more oxygen into the bulk than did OC #2.

However, the possibility exists that differences in CO<sub>2</sub> concentrations were the result of oxygen release from CeO<sub>2</sub> when at high temperature. To assess this possibility, both OCs were completely oxidized and then the gas was switched from oxidation conditions to inert conditions (part II, Figure 10). The MS data clearly show no O<sub>2</sub> signal differences between the two OCs and the baseline testing. Hence, no or inappreciable amounts of oxygen was released from CeO<sub>2</sub>, a result which is consistent with Hedayati et al's research<sup>71</sup>.

Figure 11 presents gas concentration data obtained during the reduction of the OCs within the bench-scale reactor. The data shows that CO<sub>2</sub> concentrations rose faster for both OCs, and the OCs had the ability to convert CO to CO<sub>2</sub> during the first three minutes; during the first four minutes OC #1 converted about 9.5 % more CO to CO<sub>2</sub> than did OC #2. These results suggest that OC #1 had a faster oxygen transfer rate and supplied more

oxygen for CO oxidation than did OC #2, a difference that is in agreement with the TGA-MS data.

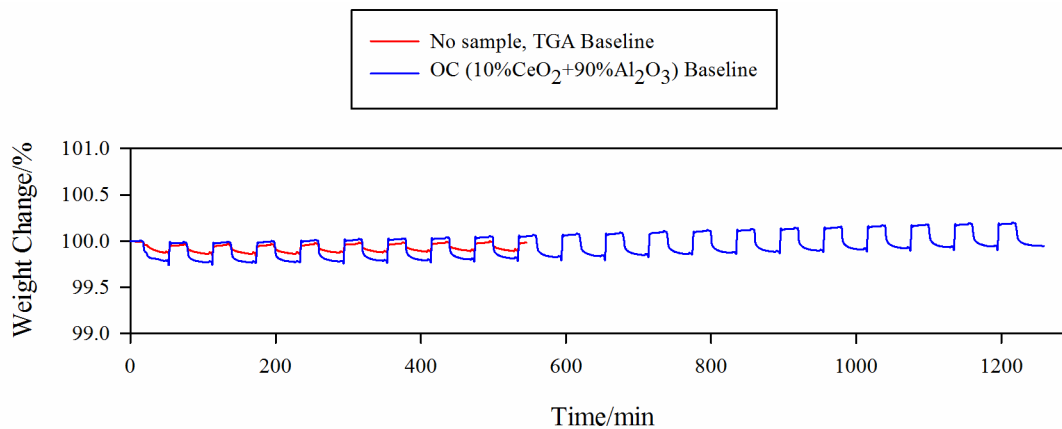


Figure 8 TGA results of baselines. The temperature was 950°C, and each redox cycle follows the sequence of purge, oxidation, purge and reduction. The reducing gas was 20 vol. % CO in argon.

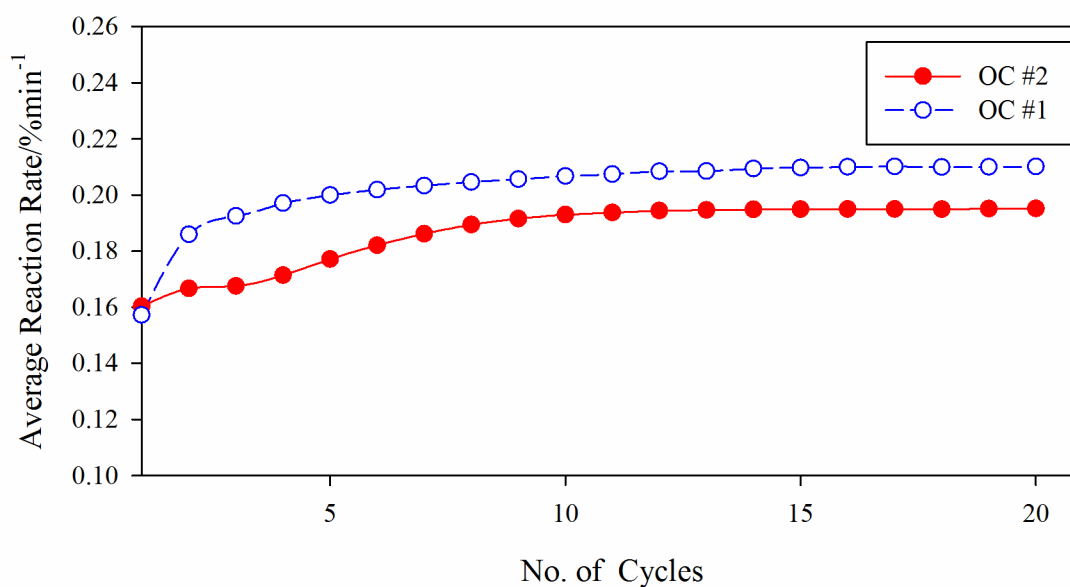


Figure 9 Comparison of average reaction rate. The temperature was 950°C, and each redox cycle follows the sequence of purge, oxidation, purge and reduction. The reducing gas was 20 vol. % CO in argon.

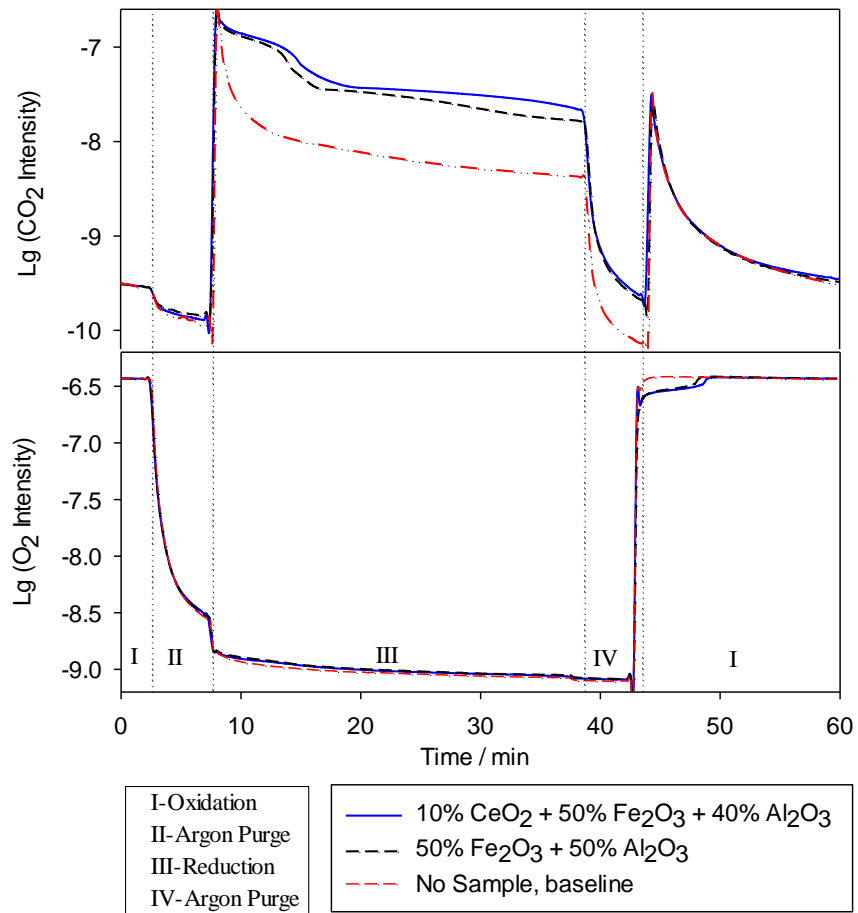


Figure 10 MS measurement, the y axis was expressed in logarithm.



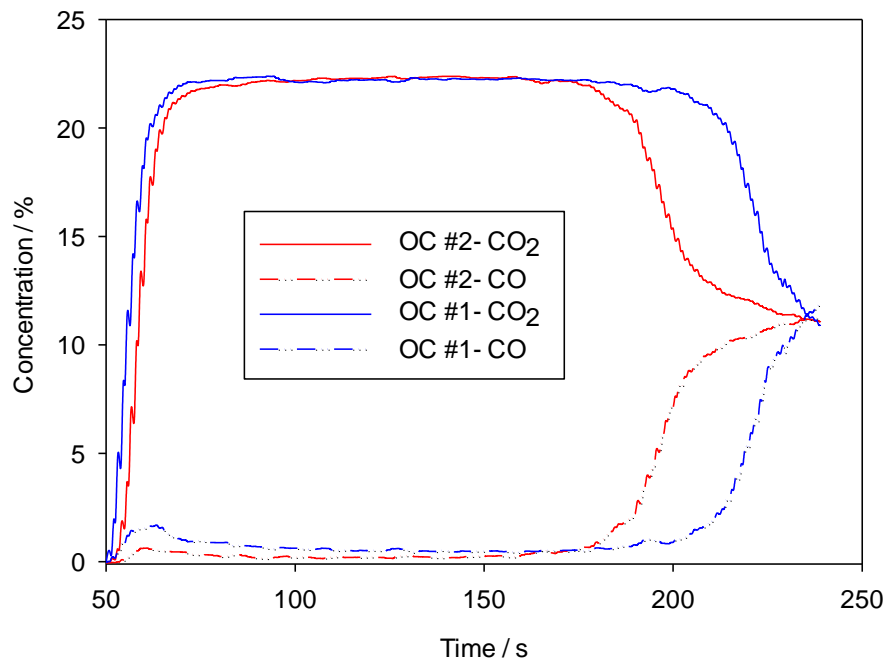


Figure 11 Gas concentrations from bench scale experiment. The temperature was 950°C, and the reducing gas was 23 vol. % CO with argon.

### 4.3 Reaction mechanism

Bi-metal oxide-based OCs usually give better performance than mono-metal oxides<sup>53</sup>, the dissimilarity of which has been attributed to “synergistic effects”. Although this terminology is generic in nature, it has entailed different mechanisms depending on the metal oxide mixtures tested. To explore possible “synergistic effects” for the CeO<sub>2</sub>-containing oxides in the current study, analytical examinations of the OCs were conducted to elucidate their chemical structure before and after exposure to oxidation and reduction conditions.

XRD data from the OCs and standard Fe<sub>2</sub>O<sub>3</sub> and CeO<sub>2</sub> are presented in Figure 12. The results for oxidized OC #1 in comparison to the reference compounds point to Fe<sub>2</sub>O<sub>3</sub> and CeO<sub>2</sub> moieties, but the OC #1 peaks are much broader which indicates a poorly crystallized or microcrystalline phase, or the presence of micro-strain<sup>91</sup>. To examine chemical structures that may provide mutual effects on the Fe<sub>2</sub>O<sub>3</sub> and CeO<sub>2</sub> crystallinity, Raman spectra were obtained from the OCs in their oxidized and reduced forms, the results of which are shown in Figure 13. The vibrational bands of OC #1 in its oxidized state are similar in position to those of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (217, 283, 398, 488, 600)<sup>92</sup> and dissimilar in intensity to the Raman bands of OC #2. This comparison suggests that CeO<sub>2</sub> has influenced the bulk phase, possibly forcing OC #1 into a highly disordered  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>-like phase.

To further examine structural differences between the OCs and possible reasons for the differences, XRD and Raman data were also acquired for reduced OC #1 (Figure 12 and Figure 13). Reduced OC #1 possesses Raman bands at 463, 608, and 694 cm<sup>-1</sup>; the sharp

band at  $463\text{ cm}^{-1}$  is similar in position to that of crystalline  $\text{CeO}_2$ <sup>93</sup>, while bands at 608 and  $694\text{ cm}^{-1}$  are not indicative of a known phase. The only iron oxide phases that would exhibit vibrational modes in the  $600\text{-}700\text{ cm}^{-1}$  region are  $\text{FeO}$  ( $650\text{ cm}^{-1}$ ) and  $\text{Fe}_3\text{O}_4$  ( $300, 520, 660\text{ cm}^{-1}$ )<sup>92</sup>. The spectrum has a small shoulder above the  $463\text{ cm}^{-1}$  band at about  $500\text{ cm}^{-1}$ , which could be related to distorted  $\text{Fe}_3\text{O}_4$ , possibly influenced by the presence of Ce in the bulk. Indeed, the XRD results show  $\text{Fe}_2\text{O}_3$  reduction creates  $\text{Fe}_3\text{O}_4$ ,  $\text{FeO}$ ,  $\text{Fe}$  and  $\text{FeCeO}_3$ . This last component is identified according to the study by Ameta et al<sup>94</sup>, and strongly suggests a solid state reaction between  $\text{Fe}_2\text{O}_3$  and  $\text{CeO}_2$  occurred during redox exposures. Therefore, it can be reasonably inferred that  $\text{Fe}_2\text{O}_3$  and  $\text{CeO}_2$  in OC #1 have formed a solid solution to some extent. Accordingly, and in consideration of: i) the  $\text{CeO}_2$  content was only 10 wt. % of OC #1; and ii) its addition significantly improved the reaction performance of the OC #1, it is hypothesized that the structure of OC #1 became a mixture of  $\text{Fe}_2\text{O}_3$  and  $\text{Fe}_2\text{O}_3\text{-CeO}_2$  in solid solution during the redox processing. As a result, the synergistic effect of  $\text{Fe}_2\text{O}_3$  with  $\text{CeO}_2$  in solid solution along with  $\text{Fe}_2\text{O}_3$  dramatically improved the reaction performance of OC #1.

It is well known that  $\text{CeO}_2$  at high temperature or in the reduced state will lose a small amount of oxygen and, thereby, generate oxygen vacancies, as defined by:



Even in pure  $\text{CeO}_2$ , oxygen vacancies have been detected<sup>93</sup>. One more important fact was that doping low valence  $\text{Fe}^{3+}$  to  $\text{CeO}_2$  will further increase the oxygen vacancy concentration. From this perspective,  $\text{Fe}_2\text{O}_3$  exerts an influence on  $\text{CeO}_2$ .

The TGA-MS results show that oxygen species on the OC surfaces are quickly consumed by fuel; however, after the surface oxygen is consumed it is necessary for diffusion of bulk oxygen to the surface if CO oxidation is to continue. This diffusion would entail a “hopping” motion of the atoms because the crystal lattice restricts the positions and migration paths of atoms <sup>95</sup>. If vacancies are generated, the least energetic path for oxygen diffusion would involve continual occupation of nearby vacancies until the oxygen is consumed at the surface. For example, data have shown that oxygen diffusion through vacancies is  $10^4$  faster than through the lattice <sup>96</sup>.

The potential of iron oxide to provide oxygen for fuel oxidation is high, about 30 % from  $\text{Fe}_2\text{O}_3$  to Fe. The rate of fuel oxidation has been the problem – low rates mean long reaction times, high OC inventory and expensive CLC processing. If oxygen from  $\text{Fe}_2\text{O}_3$  could be transferred through the oxygen vacancy generated by cerium oxides, oxygen transfer rates will be greatly increased. The TGA-MS and bench scale tests confirm that, the addition of  $\text{CeO}_2$  could help the transfer of oxygen from the bulk to the surface. The formation of  $\text{Fe}_2\text{O}_3$ - $\text{CeO}_2$  solid solution could provide this prerequisite. If so, three types of oxygen transfer could exist in this solid solution: (1) by lattice diffusion; (2) by vacancy diffusion, in which an oxygen atom could jump to a neighboring vacancy. Oxygen vacancies maybe adjacent to each other to form an oxygen transfer tunnel, so oxygen could transfer from the inside to outside only from the oxygen transfer tunnel at a very fast speed; and (3) by lattice diffusion and vacancy diffusion.

Bhavsar et al <sup>69</sup> proposed a mechanism of the interaction between  $\text{CeO}_2$  and NiO. The OC chosen by the author was NiO on the support of the reducible  $\text{CeO}_2$ . The author hypothesized the mechanism as: NiO on the surface of the OC was first reduced to

metallic Ni, then the Ni aided the cracking of CH<sub>4</sub> to C and H<sub>2</sub>, then CeO<sub>2</sub> on the metal/support interface began to provide lattice oxygen, and finally the reduced oxygen depleted cerium oxide was replenished from the core partially oxidized NiO. Our study of iron oxide-based OC with CeO<sub>2</sub> have shown that OC #1 and OC #2 had very similar performance at the beginning of the reduction reaction, whereas after this initial reactivity, the oxygen transfer rate of OC #1 became larger than that of OC #2; the implications of this reactivity are significantly different than that proposed by Bhavsar et al <sup>69</sup> because, in that case, CeO<sub>2</sub> reacted with CH<sub>4</sub> at the beginning. Furthermore, our experiments show that CeO<sub>2</sub> has very limited oxygen transfer capacity itself. However, Bhavsar et al's <sup>69</sup> oxygen transfer mechanism is also partially supported in that CeO<sub>2</sub> could transfer oxygen from a partially reduced core NiO to the surface, which in our study was related to the formation of a Fe<sub>2</sub>O<sub>3</sub> and CeO<sub>2</sub> solid solution; the solid solution should be an important step for interaction between Fe<sub>2</sub>O<sub>3</sub> and CeO<sub>2</sub>. In other words, a solid solution enables interaction of the oxygen during the reduction with not only at the surface of the metal/support interface but also throughout the whole particle.

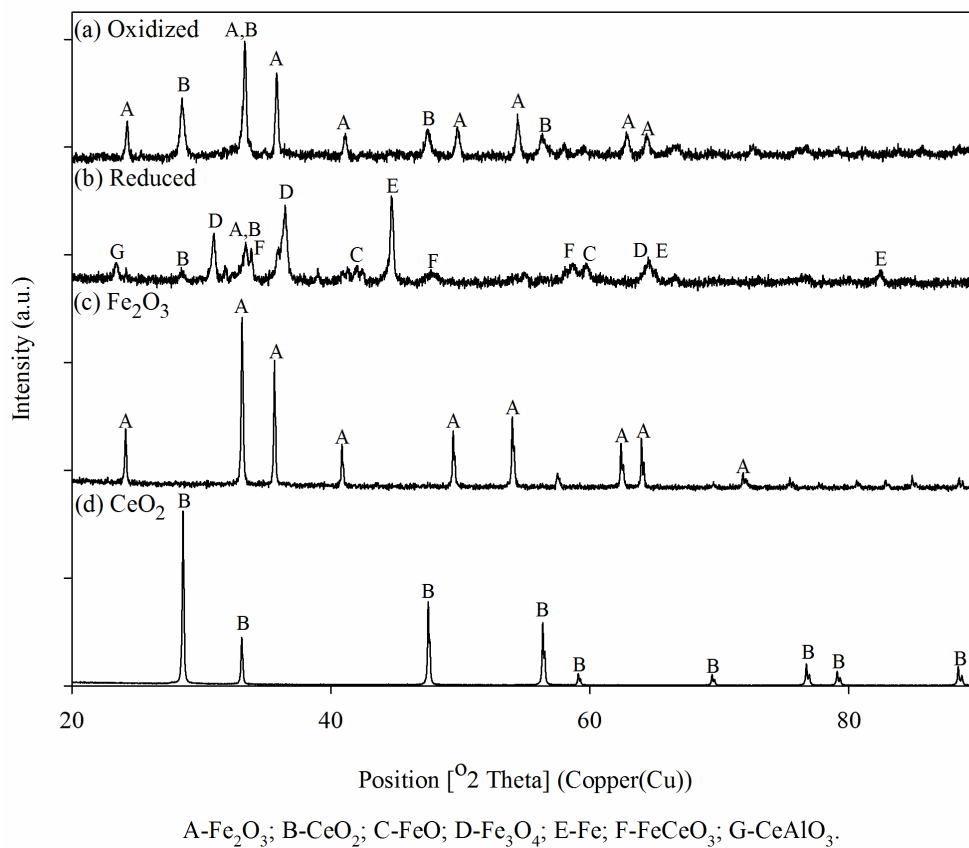


Figure 12 XRD spectra. The OC particles was crushed to fine powders before sent to the XRD test, the test was in the 2 theta range of 20°-90° with CuK $\alpha$  irradiation.

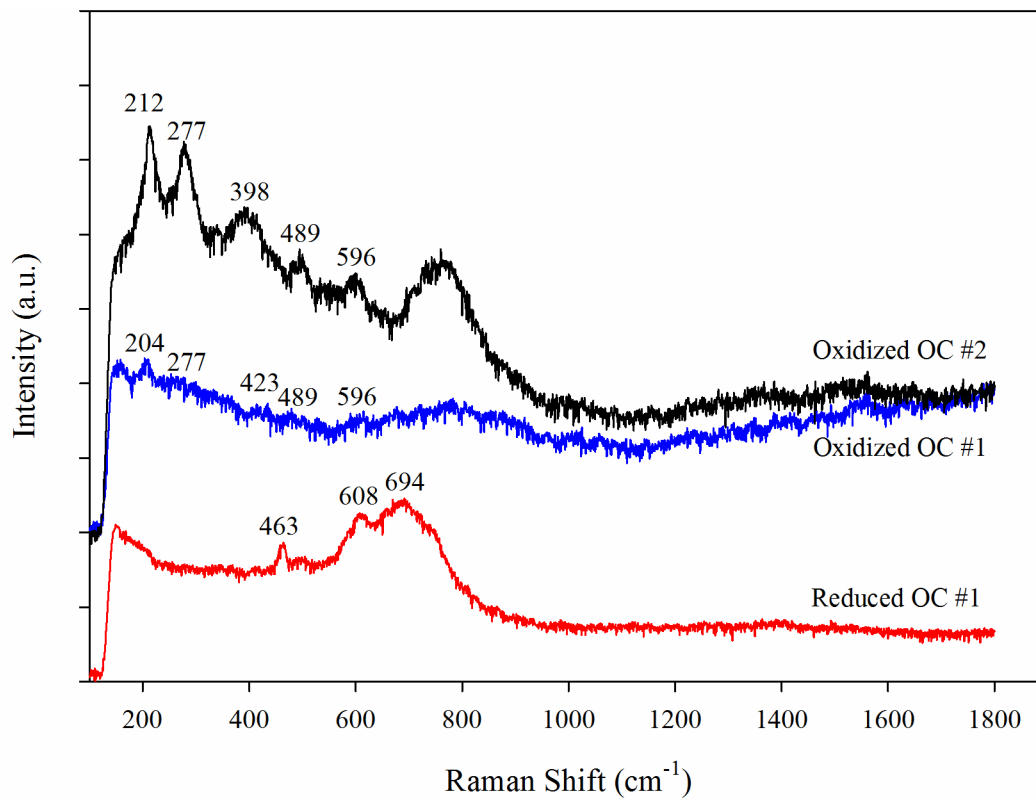


Figure 13 Raman spectra.

#### 4.4 Reaction kinetics study

##### 4.4.1 Shrinking core model (SCM)

The shrinking core model was used to carry out a kinetics analysis because a granulated structure was observed in the OC particles (Figure 5). Typically, the shrinking core model is useful in describing the reaction kinetics with this type of particle morphology<sup>35</sup>.

The basic equation for the shrinking core model can be expressed as<sup>97</sup>:

$$w = dX_r/dt = 3(1 - X_r)^{2/3}kC^n \quad (9)$$

Where the terms are defined as:  $X_r$ -conversion, in %,  $t$ -time, in s,  $k$ -reaction rate constant, for reaction order of  $n$ , the unit is in  $L^{n-1} \text{ mol}^{1-n} \text{ s}^{-1}$ ,  $C$ -reacting gas concentration, in  $\text{mol L}^{-1}$ , and  $n$ - reaction order.

Integration yields:

$$1 - (1 - X_r)^{1/3} = kC^n t \quad (10)$$

The reaction rate,  $k$ , according to Arrhenius law is expressed as:

$$k = A \exp(-E_a/RT) \quad (11)$$

Where the terms are defined as:  $A$ -pre-exponential factor, for reaction order of  $n$ , the unit is in  $L^{n-1} \text{ mol}^{1-n} \text{ s}^{-1}$ ;  $E_a$ - activation energy, in  $\text{J mol}^{-1}$ ,  $R$ -universal gas constant of  $8.314 \text{ J mol}^{-1} \text{ K}^{-1}$ , and  $T$ -temperature, in K.

##### 4.4.2 Results

The related reactions in this study are:







As stated by Abad et al <sup>98</sup>, the oxygen transferred to the fuel gas was a sum of the contribution of reduction of Fe<sub>2</sub>O<sub>3</sub>, Fe<sub>3</sub>O<sub>4</sub> and FeO, so, in agreement, this reduction kinetics study is based on the global reduction reactions shown in Equations (12) to (15). Bench scale results for OC conversion involving different temperatures and CO concentrations are shown in Figure 14 and Figure 15. The CO concentration was kept constant while the temperature was varied, and the temperature was kept constant while the CO gas concentration was varied. The relationship between conversion and time in the figures indicates that the shrinking core model was appropriate <sup>35</sup>.

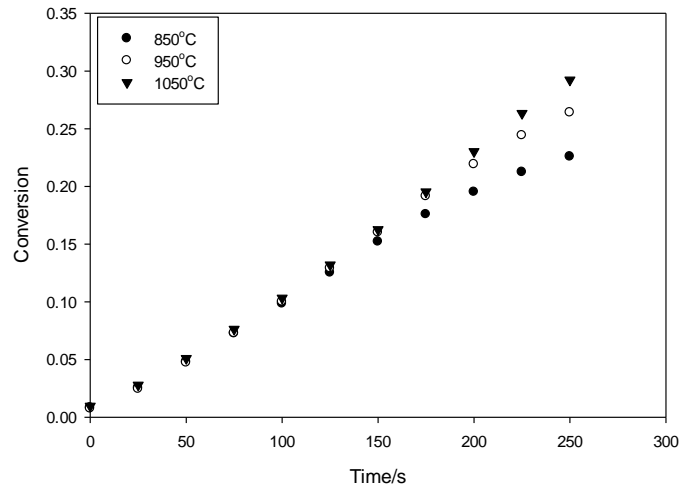
For each reaction condition, the values of  $kC^n$  (slope of the line) can be found by curve fitting. By taking the logarithm of  $kC^n$ , a straight line with a slope of  $n$  is obtained, which is the reaction order. Since the CO concentration was known, the relationship of  $\ln kC^n$  and  $\ln C$  at three different CO concentrations was plotted, as shown in Figure 16. The slope of the fitted line was found to be 0.9 for OC #1 and 1.1 for OC #2. The reaction rate constant at specific temperatures can be acquired similarly.

A transformation of Arrhenius' law can be written as:

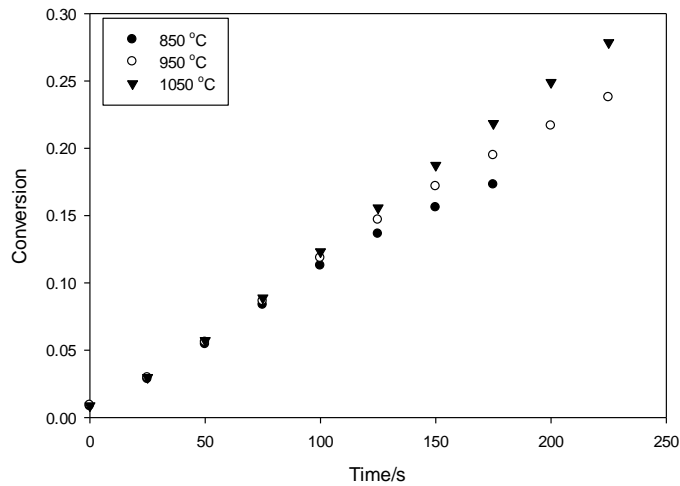
$$\ln(k) = \ln(A) + (-E_a/R)/T \quad (16)$$

By taking the Arrhenius plot, as shown in Figure 17, the activation energy and pre-exponential factors were obtained and are summarized in Table 5. The resulting kinetic parameters for the OC #1 that reacted with CO are: reaction order  $n=0.9$ , activation energy  $E_a=24.2 \times 10^3 \text{ J mol}^{-1}$  and pre-exponential factor  $A=0.006 \text{ L}^{-0.1} \text{ mol}^{0.1} \text{ s}^{-1}$ , and for OC #2 these parameters are: reaction order  $n=1.1$ , activation energy  $E_a=22.4 \times 10^3 \text{ J mol}^{-1}$  and pre-exponential factor  $A=0.003 \text{ L}^{0.1} \text{ mol}^{-0.1} \text{ s}^{-1}$ . The values for these OC#1 and OC #2

are very close, possibly because these two OCs had similar physical properties and also because kinetics studies focus on performance at the beginning of a reaction when both OCs had the same reactivity. Hence, it is probable that the differences in their reactivities after depletion of surface oxygen and the onset of faster oxygen diffusion for OC #1 was not measured in these kinetics studies.

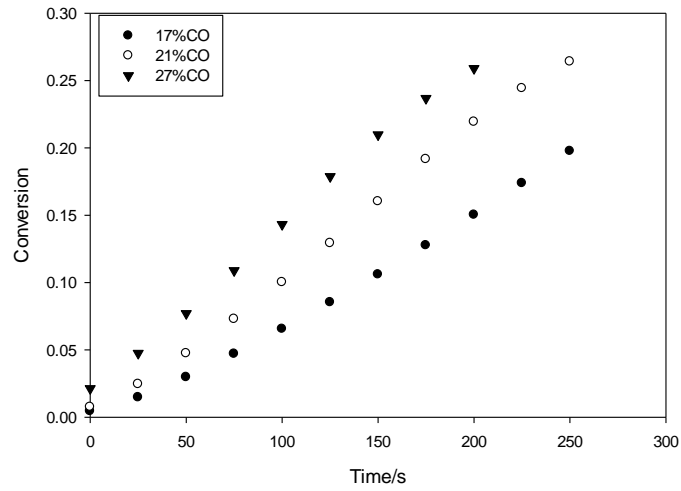


(a) OC #1

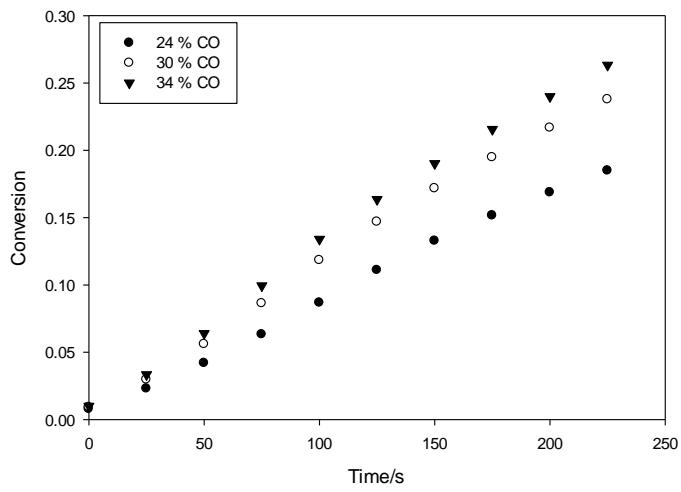


(b) OC #2

Figure 14 Temperature effect on conversion

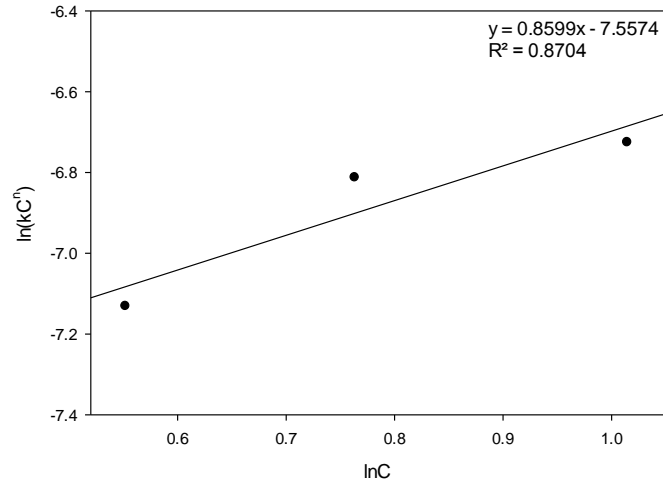


(a). OC #1

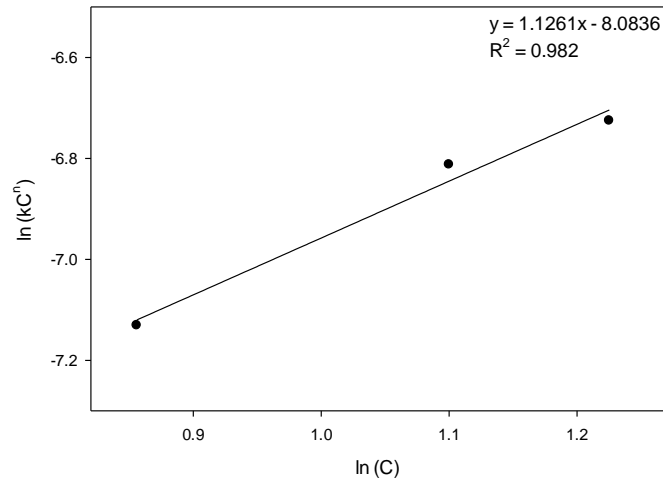


(b). OC #2

Figure 15 CO concentration effect on conversion

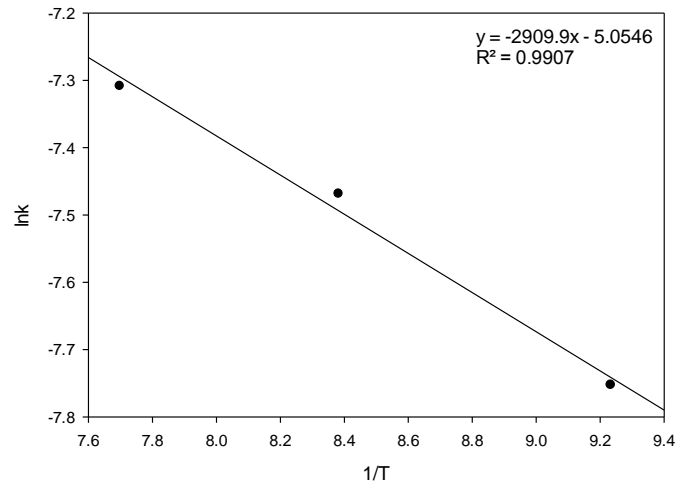


(a). OC #1

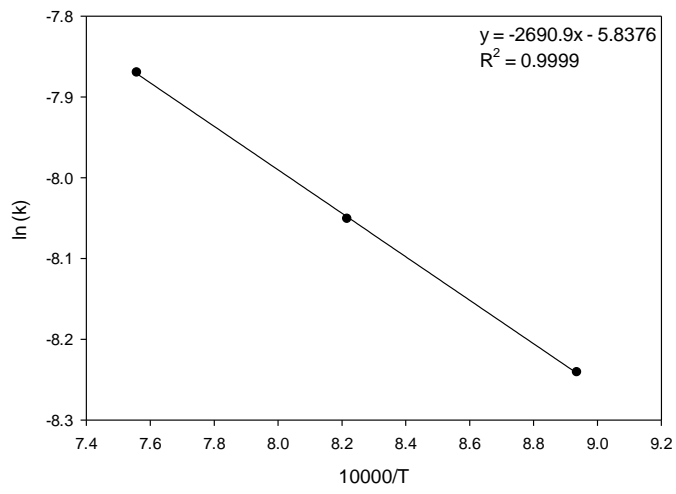


(b). OC #2

Figure 16 Plot of  $\ln(kC^n)$  and  $\ln(C)$



(a). OC #1



(b). OC #2

Figure 17 Arrhenius plot

Table 5 Kinetic parameters

Kinetic parameters	$E_a$	$A$	$n$
Unit	J mol <sup>-1</sup>	L <sup>n-1</sup> mol <sup>1-n</sup> s <sup>-1</sup>	-
OC #1	24.2	0.006	0.9
OC #2	22.4	0.003	1.1

#### 4.5 Conclusions

Iron-based OCs with and without CeO<sub>2</sub> additive were produced and physical properties such as morphology, surface area, and mechanical strength were investigated. The reactivity was tested by both TGA-MS and bench scale methods. The results for both OCs were compared and analyzed and the promotional role of CeO<sub>2</sub> was discussed. The following conclusions were drawn: (1) The addition of CeO<sub>2</sub> has no effect on the OCs morphology, BET surface area, or the mechanical strength; (2) Although OC #1 had higher reactivity than OC #2, this difference was not caused by differences in BET surface area or additional oxygen release from CeO<sub>2</sub> by itself; (3) The proposed promotional role of the CeO<sub>2</sub> is that it enables the creation of oxygen vacancies in a solid solution. These vacancies were able to transfer oxygen from Fe<sub>2</sub>O<sub>3</sub> quickly to the surface of the OC by vacancy diffusion or through oxygen tunnel transfer. The formation of a CeO<sub>2</sub> and Fe<sub>2</sub>O<sub>3</sub> solid solution provides the prerequisite for these short range interactions. (4) The kinetic parameters of both OCs are close to each other.

## 5. Scale-up Study

### 5.1 Governing equations

Mathematical modeling is the most basic approach to scale-up<sup>81</sup>. Anderson and Jackson<sup>99</sup> derived the governing equations for fluidized beds particles, these equations are as follows:

Global continuity equation:

$$\varepsilon_g + \varepsilon_s = 1 \quad (17)$$

Gas-phase continuity equation:

$$\frac{\partial \varepsilon_g}{\partial t} + \frac{\partial}{\partial x_i} (\varepsilon_g V_{gi}) = 0 \quad (18)$$

Solid-phase continuity equation:

$$\frac{\partial \varepsilon_s}{\partial t} + \frac{\partial}{\partial x_i} (\varepsilon_s V_{si}) = 0 \quad (19)$$

Gas-phase momentum equation:

$$\rho_g \varepsilon_g \left[ \frac{\partial V_{gi}}{\partial t} + V_{gj} \frac{\partial V_{gi}}{\partial x_j} \right] = -\frac{\partial P}{\partial x_i} - \varepsilon_g \frac{\partial \tau_{ij}}{\partial x_j} - \beta (V_{gi} - V_{si}) \quad (20)$$

Solid-phase momentum equation:

$$\rho_s \varepsilon_s \left[ \frac{\partial V_{sj}}{\partial t} + V_{sj} \frac{\partial V_{sj}}{\partial x_j} \right] = -\varepsilon_s \frac{\partial \sigma_{ij}}{\partial x_j} + \rho_s \varepsilon_s g_i + \beta (V_{gi} - V_{si}) \quad (21)$$

These equations can describe particles' movement very well, but they are too complicated to solve even with present computational means<sup>81</sup>. To successfully implement a scaling study it is therefore necessary to simplify these equations, one approach of which is to invoke hydrodynamic and chemical similarity by using sets of dimensionless numbers which have to be kept constant at both scales<sup>87</sup>.

Glicksman<sup>100</sup> simplified the above governing equations and derived the scaling numbers by non-dimensionalizing these governing equations. They are called full set scaling laws and are as follows:

$$\frac{u_o \rho_g d_p}{\mu}, \frac{u_o^2}{gD}, \frac{\rho_g}{\rho_p}, \frac{D}{L}, \frac{d_p}{D}, \varphi; \Phi \quad (22)$$

However, using the full set of Glicksman scaling laws brings many constraints and is very difficult to implement<sup>87</sup>. To simplify the scaling laws and to make it easier to be implemented, we made the following assumptions: the governing forces in fluidized bed are inertia and buoyancy, and the viscous force is ignored, so viscosity is not important. Thus, it is reasonable to drop the Reynolds number. Furthermore, because this scale-up study is based on a hot model, it is necessary to add a dimensionless number reflecting the chemical reaction. The Damköhler number is a ratio of reaction rate to transport rate. It is an important parameter for chemical reaction, so the Damköhler number was added to the scaling laws. As a consequence, the modified scaling laws are proposed as follows:

$$\frac{u_o^2}{gD}, \frac{\rho_g}{\rho_p}, \frac{D}{L}, \varphi; \Phi, Da \quad (23)$$

The simplified scaling laws eliminate some constraints and give better flexibility in model design, which provide freedom to choose the diameter of the cold flow model without a need for exotic particles and pressures. Hence, in this study, the same oxygen carrier (OC #1) was used for both reactors. As mentioned before, the OC has a composition of 50 wt. % Fe<sub>2</sub>O<sub>3</sub>, 10 wt. % CeO<sub>2</sub> and 40 wt. % Al<sub>2</sub>O<sub>3</sub>, with a size range of 150-300 μm.

Improvement in the potential of combustion scaling is possible if important parameters can be kept identical in a test plant and in a full-scale plant to meet combustion similarity.



These parameters include bed temperature, total excess-air ratios, primary stoichiometry, fuel type, and bed material<sup>81, 101</sup>. An important parameter for combustion similarity is reactor performance as measured by fuel conversion, product distribution, temperature distribution and selectivity<sup>102-104</sup>. For this study, reactors were designed and the operating parameters chosen based on the simplified set of scaling laws; reactor performance in a hot model was the focus. Hot model performance can be used to reflect the fluidization state. Conversion and temperature changes during the reactor testing were chosen for validating the scaling laws.

## 5.2 Experimental

An important aspect of model experiments is to confirm the similarity between the original phenomena and the scale model counterpart<sup>84, 105</sup>. In this study, the bench scale CLC setup was modified to conduct experiments.

Two reactors, reactor-1 and reactor-2, were fabricated with the scale ratio  $r = 2$ , as shown in Figure 18. Similar to the simplified fluidized bed scaling laws, to build these two reactors, Horio<sup>106</sup> suggested that:

$$r = \frac{D^l}{D^s} = \frac{L^l}{L^s} \quad (24)$$

$$\sqrt{r} = \frac{u_{mf}^l}{u_{mf}^s} = \frac{(u_o - u_{mf})^l}{(u_o - u_{mf})^s} \quad (25)$$

Hence, the reactors were built to meet hydrodynamic similarity. They were operated at the same temperature and pressure and, as mentioned previously, the materials in the two reactors were from the same batch. In this study, the Da was defined as:

$$Da = \frac{k_v}{u_o/L} \quad (26)$$

It scales with  $\sqrt{r}$ . For scaling validation, the time in the two different reactors must be scaled by a factor of  $\sqrt{r}$ <sup>87</sup>, namely:

$$\sqrt{r} = \frac{Da^l}{Da^s} \quad (27)$$

$$\sqrt{r} = \frac{t^l}{t^s} \quad (28)$$

Based on the above equations, the reactors design and operating parameters were determined and are shown in Table 6.

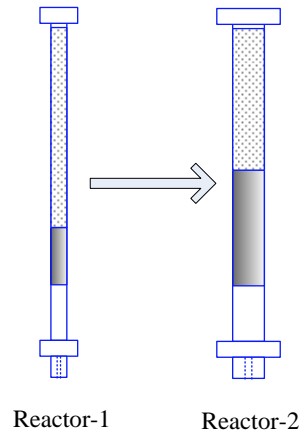


Figure 18 Two reactors.

Table 6 Reactor design and operate parameters

	$D$	$L$	$M$	$Q$	$T$	$P$	$u_{mf}$	$k_v$	$Da$
Units	mm	mm	g	L/min	°C	bar	cm/s	s <sup>-1</sup>	-
Reactor-1	25.4	3.81	12.5	2.8	970	1.0	2.6	0.00517	0.0022
Reactor-2	50.8	7.62	100	16	970	1.0	4.5	0.00520	0.0030

### 5.3 Results and discussion

In this study, the validation parameters were monitored by using the LabView program and by recording data every 0.5 seconds were the fuel conversion during reduction and temperature changes. The conversion  $\chi$  was calculated by:

$$\chi = \frac{C_{CO_2}}{C_{CO} + C_{CO_2}} \quad (29)$$

These  $\chi$  values-versus-time, presented in Figure 19 for both reactors, were very close to each other. At the beginning of the reaction, the OC induced high conversion because ample amounts of surface oxygen were available to convert CO into CO<sub>2</sub>. However, as the reaction proceeded, the amount of oxygen available from the OC became less and less and led to a decrease in conversion. The conversion values decreased because oxygen was being supplied from the bulk of the OC, a process slower than oxygen transfer from the surface. Hence, for these reactors at a time greater than about 25 seconds, the conversion was controlled by oxygen diffusion from the bulk to the surface of the OC. Overall, the close match of conversion between reactor 1 and reactor 2 validates the scaling laws that were used.

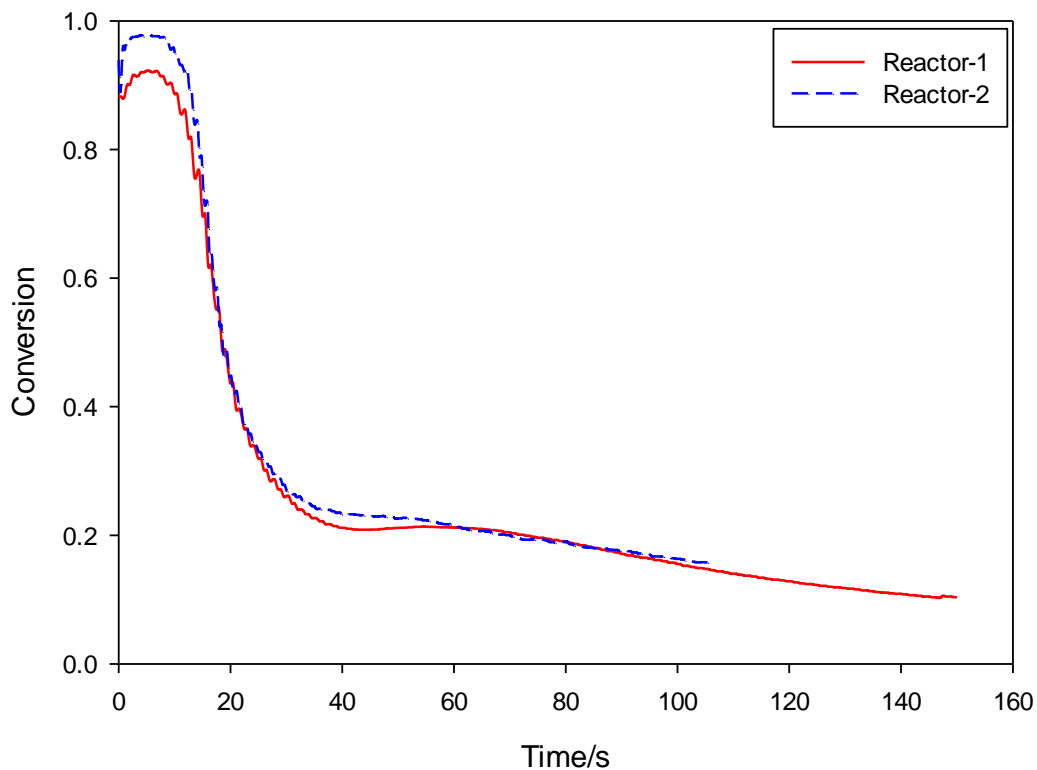


Figure 19 CO conversion.

Figure 20 shows the trends in temperature changes during the reduction reaction within the two reactors were very similar. Although not exactly identical, the temperature differences between the two reactors were very small compared to high temperature used during the testing. Temperature increases were noticed for both reactors because the reactions were exothermic, as shown in Table 3. Hence, the data confirms that appropriate scaling laws were used.

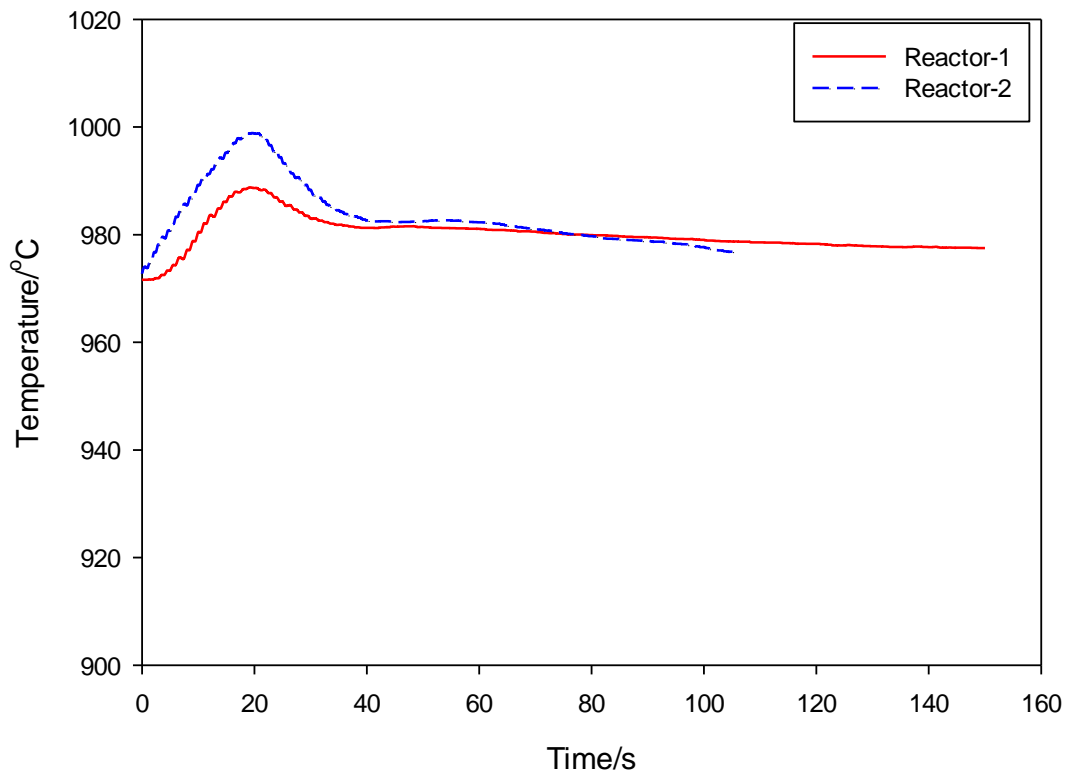


Figure 20 Temperature change.

#### 5.4 Conclusions

Scaling laws of fluidized bed were applied to CLC reactors. Two CLC reactors with different size were built, as guided by scaling laws to meet hydrodynamic similarity, and then tested. Combustion similarity was the focus of this study and was monitored by acquiring and analyzing fuel conversion and temperature changes during the CLC reaction process. Comparison of conversion and temperature changes showed excellent combustion similarities in the two reactors, thereby validating the scale modeling method and the scale laws for CLC in fluidized beds. These results suggest the possibility for using scaling laws to scale-up even larger CLC reactors in the future.

## 6. Conclusions and Future Work

There is a general consensus on the need to reduce emissions of CO<sub>2</sub> from industrial process to limit or slow climate change resulting from increasing greenhouse gas concentrations in Earth's atmosphere. CLC has the advantages of in-situ oxygen separation, low NO<sub>x</sub> emissions and low cost of CO<sub>2</sub> emission abatement. The use of CLC for power generation is an advanced energy technology that can capture CO<sub>2</sub> inherently, which could prove to be the next electricity generation technology in a carbon constrained future.

OC development and reactor scale-up are two important tasks for CLC technologies. Iron oxide based OCs, although suitable for CLC application, have less than optimum reactivity but embody attributes attractive for CLC and, hence, have attracted considerable research attention in recent years. Bi-metal oxide OCs usually impart better CLC performance than mono-metal oxide OCs. As a consequence, this dissertation focused on using CeO<sub>2</sub> as an additive to iron oxides based OCs to attempt to further improve reactivities.

Iron-based OCs with and without CeO<sub>2</sub> additive were produced and studied. The addition of CeO<sub>2</sub> had no effect on the OCs' morphology, BET surface area, or the mechanical strength. The reactivity at the surface of both OCs was close, and the kinetic parameters for both OCs were similar. Beyond the surface, OC with CeO<sub>2</sub> additive showed higher reactivity than OC without CeO<sub>2</sub>. This difference was not caused by the BET surface area or an additional oxygen release from CeO<sub>2</sub> by itself. The proposed promotional role of CeO<sub>2</sub> additive is that it formed a solid solution with the iron oxide which enabled more

efficient diffusion through vacancy hopping or tunneling of bulk oxygen to the metal oxide surface where reduction and oxidation reactions occurred. The formation of the  $\text{CeO}_2\text{-Fe}_2\text{O}_3$  solid solution was a prerequisite for improving the reactivity of the iron oxide OC.

Two CLC reactors with different size were built, guided by scaling laws to meet hydrodynamic similarity. Combustion similarity, fuel conversion and temperature change, during the CLC reaction process was established. Comparison of these factors showed excellent similarities in the two reactors, thereby validating the scale modeling method and the scale laws for CLC in fluidized beds. These results suggest the possibility for using scaling laws to scale-up even larger CLC reactors in the future.

In the future, several research directions need to be further studied. One direction is to test the  $\text{CeO}_2$  promoted OC toward solid fuels because solid fuels such as coal can be used as fuel in CLC, and not enough gaseous fuels are available for CLC to meet potential commercial power generation needs. Coal is still the main energy source for power generation in the U.S.A and is also a dominant fuel in the world. Furthermore, coal deposits are well understood and abundant, making coal lower price option for energy generation. Thus, it is necessary to study the performance of  $\text{CeO}_2$  promoted OCs with coal. Coals from difference places, like Powder River Basin (PRB), Illinois and Kentucky have different compositions and their reactivities to CLC may also be different.

The second direction is to study the water vapor effect on the performance of OCs. When solid fuel is used as fuel, water vapor is usually used as the fluidizing gas, because water vapor will gasify the solid fuel to gaseous fuel (i.e. coal gasification with water vapor to

produce H<sub>2</sub> and CO). The emergence of water vapor will affect the OCs' performance. The effect of H<sub>2</sub>O (v) needs to be figured out.

The third direction is to continue reactor scale-up studies. The ultimate goal of CLC research is to apply CLC technology for power generation, thus commercial scale CLC technologies will need to be studied. Starting from the bench scale reactor, and through the scale modeling method, it may be possible to scale up to pilot scale reactors, and then the pilot scale reactors can be further scaled up to commercial scale, during which time process improvements and sophistication may need to be added.



## Nomenclature

<i>A</i>	Pre-exponential factor; for reaction order of $n$ , the unit is $\text{L}^{n-1} \text{mol}^{1-n} \text{s}^{-1}$
<i>C</i>	Gas concentration; $\text{mol L}^{-1}$
<i>D</i>	Diameter, m
<i>Da</i>	Damköhler number
<i>E<sub>a</sub></i>	Activation energy, $\text{J mol}^{-1}$
<i>g</i>	Gravity, $9.8 \text{ m s}^{-2}$
<i>H</i>	Heat of formation, $\text{J mol}^{-1}$
<i>L</i>	Bed height, m
<i>k</i>	Reaction rate constant; for reaction order of $n$ , the unit is $\text{L}^{n-1} \text{mol}^{1-n} \text{s}^{-1}$
<i>m</i>	Mass, grams
<i>n</i>	Reaction order
<i>P</i>	Pressure, pa
<i>Q</i>	Flow rate, $\text{L min}^{-1}$
<i>R</i>	Universal gas constant, $8.314 \text{ J mol}^{-1} \text{ K}^{-1}$
<i>R<sub>o</sub></i>	Oxygen transport capacity; %
<i>r</i>	Scale ratio

$T$	Temperature, K
$t$	Time, s
$u$	Velocity, m s <sup>-1</sup>
$V$	Vector velocity
$w$	Reaction rate, % s <sup>-1</sup>
$X$	Conversion of reduction, %
$\rho$	Density, kg m <sup>-3</sup>
$\Delta H$	Heat of reaction/combustion, J mol <sup>-1</sup>
$v_i'$	Mole concentration coefficient of reactant
$v_i''$	Mole concentration coefficient of product
$\tau_{ij}$	Gas phase stress tensor
$\sigma_{ij}$	Solid phase stress tensor
$\varepsilon$	Volume fraction, %
$\mu$	Dynamic viscosity, pa·s
$\beta$	Drag coefficient
$\varphi$	Sphericity
$\Phi$	Particle size distribution
$\varepsilon_H$	Voidage

$\chi$  Conversion, %

### Subscripts

g Gas phase

mf Minimum fluidization

s Solid phase

o Superficial

ox Oxidized state

red Reduced state

p Particle

r Reduction

### Superscripts

*l* Big

*o* Standard temperature, 298K

*s* Small

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#### Publications

##### Journals:

**Liu, F.**; Zhang, Y.; Chen, L.; Qian, D.; Neathery, J.; Saito, K.; Liu, K.; "Investigation of a Canadian Ilmenite as Oxygen Carrier for Chemical Looping Combustion"; Energy & Fuel, 2013. **27**(10): p. 5987-5995.

**Liu, F.**; Saito, K.; Liu K.; "Scale-up of Chemical Looping Combustion"; Springer, Progress in Scale Modeling, volume 2, 2014.

**Liu, F.**; Neathery, J.; Saito, K; Liu, K.; "CeO<sub>2</sub> Promoted Iron Based Oxygen Carrier for Chemical Looping Combustion"; submitted to the 35<sup>th</sup> International Symposium on Combustion.

Zhang, Y.; **Liu, F.**; Chen, L.; Han, C.; Neathery, N.; Liu, K.; " Investigation of Water Vapor Influence on the Performance of Iron, Copper and Nickel Based Oxygen Carriers for Chemical Looping Combustion", Energy & Fuel; 2013. **27**(9): p. 5341-5351.

Chen, L.; Zhang, Y.; **Liu, F.**; Neathery, N.; Liu, K.; "Development of cost-effective oxygen carrier from red mud for direct-coal-fueled chemical-looping combustion", submitted to Energy & Fuel.

**Presentations/Proceedings:**

**Liu, F.**; Li, T.; Neathery, J.; Liu, K.; Saito, K. "Characterization and Kinetic Study of Ilmenite for Chemical Looping Combustion," CSSCI (Central States Section of the Combustion Institute) April 22-24, Dayton, OH, 2012

**Liu, F.**; Zhang, Y.; Rubel, A; Neathery, J.; Liu, K. "Density, Particle Size Based Ash Separation from Oxygen Carrier and Char in Bench Scale Chemical Looping Combustion System"; 10th Annual Conference on Carbon Capture & Sequestration, May 2-5, Pittsburgh, Pennsylvania, 2011.

**Liu, F.**; Saito, K.; Liu K; "CO<sub>2</sub> Capture and Possible Impact on Painting"; Painting Technology Workshop, October 30-31, Lexington, Kentucky, 2012

Slone, J.; **Liu, F.**; Zhang, Y.; Rubel, A.; Neathery, J.; Liu, K.; " Characterization of Ilmenite for Chemical-Looping Combustion by Carbon Monoxide in a Bench-Scale

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Chen, L.; Zhang, Y.; **Liu, F.**; Han, C.; Neathery, J.; Liu, K.; " Development and Performance Evaluation of a Cost-effective Oxygen Carriers from Red Mud for Chemical Looping Combustion" 2012 AIChE Annual meeting, Oct 28–Nov 2, Pittsburgh, Pennsylvania, 2012