



Enhanced thermal rectification in graded $\text{Si}_c\text{Ge}_{1-c}$ alloys

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ABSTRACT

We evaluate the rectification coefficient of the heat flux in graded $\text{Si}_c\text{Ge}_{1-c}$ alloys, both for longitudinal heat flow along a direction z and radial heat flow along a direction r , and for linear stoichiometric spatial distributions $c(z)$ and $c(r)$, with $c \in [0, 1]$. To this end, we take an expression for the thermal conductivity $\lambda(c, T)$ fitting the experimental data in the whole interval $0 \leq c \leq 1$, for different values of the temperature T . Such an expression leads to a highly nonlinear evolution equation for the temperature. It turns out that the systems under consideration are very efficient as thermal rectifier in the full stoichiometric range, and can be used as thermal diode.

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1. Introduction

Graded Si/Ge alloys, with composition c changing along the length of the system, have many practical applications in heat transfer problems [1–6]. One of them is heat rectification, namely, the fact that a same difference of temperature but applied in opposite verses along the same direction, generates different heat flows. Such a possibility represents one of the basic features of phononics [7–10], as it provides the theoretical basis for the construction of phononic diodes. This phenomenon arises, for instance, in binary alloys with thermal conductivity depending on a stoichiometric variable c related to the concentration of the different constituents. Thus, thermal rectification is expected in systems changing their composition as, for instance, $\text{Si}_c\text{Ge}_{1-c}$ [11–15], with the stoichiometric composition parameter c gradually changing as a function of the position. The aim of this paper is to analyze in deeper detail heat rectification in $\text{Si}_c\text{Ge}_{1-c}$ nanowires. Hence, first we take an expression for the thermal conductivity $\lambda(c, T)$ fitting the experimental data in the whole interval $0 \leq c \leq 1$ at several temperatures, which leads to a highly nonlinear evolution equation for the temperature. We explore the consequences of the variable composition on the value of the rectification coefficient $R[c, T_H, T_C]$, with T_H and T_C the hottest and coldest temperatures, respectively, applied to the ends of the nanowire, and R defined as the ratio of the absolute value q_r of the reverse heat flux with respect to a

given reference verse, over the absolute value q_d of the direct heat flux with respect to the same verse, i.e., $R[c, T_H, T_C] \equiv q_r/q_d$ [7–10], for given values of T_H and T_C and for a given spatial distribution c . We consider this phenomenon not only for longitudinal heat flow along a nanowire but also for radial heat flow across a cylindrical device with internal radius r_{int} and external radius r_{ext} . In this situation, the material inhomogeneity in the composition is superposed to a geometrical inhomogeneity in the transversal area of the system. We explore the combined consequences of both factors on the value of the rectification coefficient. The paper presents the following layout. In Section 2, we derive a constitutive equation for the thermal conductivity $\lambda(c, T)$ of $\text{Si}_c\text{Ge}_{1-c}$ alloys based on experimental data. In Section 3, we solve the stationary temperature equation in two one-dimensional systems of length $L = 100$ nm and $L = 30$ nm, respectively, and use this solution to obtain the values of the rectification coefficient for different values of the applied heat flux. In Section 4, we consider two cylindrical systems, whose section is constituted by an annulus of radius $R_a \equiv r_{ext} - r_{int} = 100$ nm, and $R_a \equiv r_{ext} - r_{int} = 30$ nm, with r_{ext} and r_{int} the external and internal radius, respectively. For such systems, under the hypothesis of radial heat flow along the direction r , we calculate the corresponding heat rectification coefficient. In Section 5, we discuss the results and their possible extensions.

2. Thermal conductivity of $\text{Si}_c\text{Ge}_{1-c}$ alloys

2.1. Best fit of thermal conductivity

Let us consider a $\text{Si}_c\text{Ge}_{1-c}$ alloy, whose composition c changes along a given direction z . For the sake of illustration, in Table 1 we

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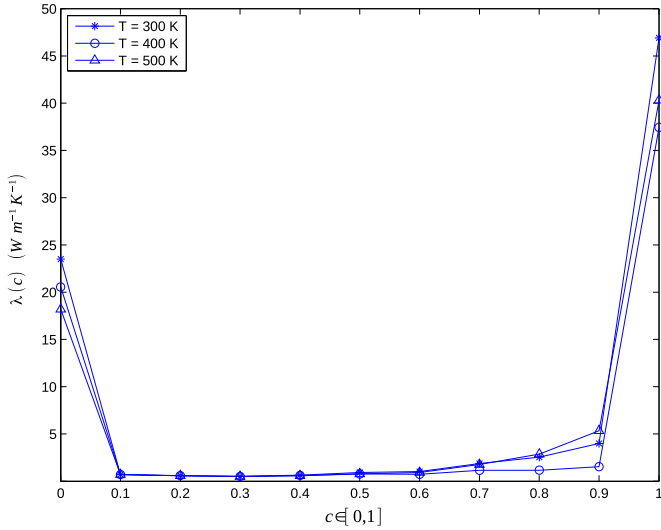


Fig. 1. Sketch of thermal conductivity of a nanowire of a $\text{Si}_c\text{Ge}_{1-c}$ nanowire of length $L = 100$ nm in terms of c , at $T = 300$ K, $T = 400$ K, and $T = 500$ K.

Table 1

Values of the thermal conductivity (in $\text{Wm}^{-1}\text{K}^{-1}$) corresponding to the mentioned compositions at $T = 300$ K, $T = 400$ K and $T = 500$ K, for bulk systems, (from [16–19]).

	λ_{Si}	$\lambda_{\text{Si}_{0.9}\text{Ge}_{0.1}}$	$\lambda_{\text{Si}_{0.1}\text{Ge}_{0.9}}$	λ_{Ge}
$T = 300$ K	149.95	25.13	10.46	77.95
$T = 400$ K	113.54	23.90	10.06	59.42
$T = 500$ K	92.01	22.79	9.69	48.08

Table 2

Values of the parameters in the analytical expression (1) of thermal conductivity in terms of the composition for a $\text{Si}_c\text{Ge}_{1-c}$ alloy, $L = 100$ nm [21].

	A	B	D	E	F	G
$T = 300$ K	1.348	6.38	-5.363	22.145	252.53	-251.94
$T = 400$ K	1.331	6.305	-5.282	19.181	239.73	-239.15
$T = 500$ K	1.309	6.208	-5.1836	16.851	228.19	-227.63

give the values of the thermal conductivity for four compositions and three temperatures.

In Fig. 1 it is shown the plot of the function $\lambda(c, T)$ for $\text{Si}_c\text{Ge}_{1-c}$ at $T = 300$ K (*), $T = 400$ K (○), and $T = 500$ K (△), for $L = 100$ nm, which has been obtained in [4] by using the results in [16–18].

The variation of λ with c is steepest in the ranges $0 \leq c \leq 0.1$ and $0.9 \leq c \leq 1$ since a small amount of impurities of different mass strongly contributes to phonon scattering, thus reducing very much thermal conductivity as compared with the corresponding pure system [20]. In Ref. [21] it has been proved that the behavior above of the heat conductivity $\lambda(c, T)$, as a function of the composition c and the temperature T , can be expressed analytically as sum of exponential functions. In particular, the following constitutive equation for $\lambda(c, T)$ was derived:

$$\lambda(c) = Ae^{Bc^2+Dc} + Ee^{Fc^2+Gc} \quad (1)$$

where A, B, D, E, F, G , are suitable temperature-dependent parameters whose values at the constant temperatures $T = 300$ K, $T = 400$ K, $T = 500$ K, are shown in Table 2 and in Table 3 (for $\text{Si}_c\text{Ge}_{1-c}$ and $\text{Si}_{1-c}\text{Ge}_c$ nanowires, respectively, of length $L = 100$ nm) and in Table 4 and in Table 5 (for $\text{Si}_c\text{Ge}_{1-c}$ and $\text{Si}_{1-c}\text{Ge}_c$ nanowires, respectively, of length $L = 30$ nm).

Table 3

Values of the parameters in the analytical expression (1) of thermal conductivity in terms of the composition for a $\text{Si}_{1-c}\text{Ge}_c$ alloy, $L = 100$ nm [21].

	A	B	D	E	F	G
$T = 300$ K	3.73	6.38	-7.3978	39.858	252.524	-253.11
$T = 400$ K	3.70	6.305	-7.327	34.465	239.737	-240.32
$T = 500$ K	3.64	6.207	-7.232	29.761	228.197	-228.77

Table 4

Values of the parameters in the analytical expression (1) of thermal conductivity in terms of the composition for a $\text{Si}_c\text{Ge}_{1-c}$ alloy, $L = 30$ nm [21].

	A	B	D	E	F	G
$T = 300$ K	0.92	6.285	-5.28	12.07	229.09	-228.44
$T = 400$ K	0.897	6.21	-5.18	10.98	213.04	-212.48
$T = 500$ K	0.888	6.15	-5.12	10.17	206.72	-206.17

Table 5

Values of the parameters in the analytical expression (1) of thermal conductivity in terms of the composition for a $\text{Si}_{1-c}\text{Ge}_c$ alloy, $L = 30$ nm [21].

	A	B	D	E	F	G
$T = 300$ K	2.50	6.285	-7.29	22.96	229.09	-229.73
$T = 400$ K	2.51	6.21	-7.24	19.18	213.04	-213.59
$T = 500$ K	2.49	6.15	-7.18	17.65	206.73	-207.28

Table 6

Values of the coefficients a_i, b_i, d_i, e_i, f_i and g_i for a $\text{Si}_c\text{Ge}_{1-c}$ alloy, $L = 100$ nm.

a_1	a_2	b_1	b_2
-0.000145	-2.5×10^{-7}	-0.00064	-1.1×10^{-6}
d_1	d_2	e_1	e_2
0.000723	8.7×10^{-7}	-0.03281	0.0000317
f_1	f_2	g_1	g_2
-0.1343	0.000063	0.13425	-0.0000635

2.2. Temperature dependence of material parameters

As far as the dependence of material parameters on the temperature, we take a series expansion of them around the room temperature, namely $T = 300$ K, up to the second order in T . For the sake of illustration, we show the procedure for $A(T)$ in the case of a $\text{Si}_c\text{Ge}_{1-c}$ alloy, with $L = 100$ nm. We write

$$A(T) = A(300) + a_1(T - 300) + a_2(T - 300)^2 \quad (2)$$

In this way, taking into account the values of A in Table 2, we get

$$A(400) = 1.348 + a_1(100) + a_2(100)^2 = 1.331 \quad (3)$$

$$A(500) = 1.348 + a_1(200) + a_2(200)^2 = 1.309 \quad (4)$$

The linear system (3)–(4) allows the determination of the coefficients a_1 and a_2 . Such a procedure can be repeated for the parameters B, D, E, F , and G too, both for $L = 100$ nm and $L = 30$ nm, and both for $\text{Si}_c\text{Ge}_{1-c}$ and $\text{Si}_{1-c}\text{Ge}_c$. The results are shown in the Tables from 6 to 9. In these tables, b_i, d_i, e_i, f_i and g_i respectively correspond to the expansion of the coefficients B, D, E, F , and G in terms of T analogous to a_i in the expansion (2) of coefficient $A(T)$.

In this way we can express the thermal conductivity as function of c and T for all the systems of interest. It is worth observing that the procedure illustrated so far does not depend on the geometry of the system, and hence the fit in (1) can be applied, with the same values of the parameters A, B, D, E, F , and G determined above, to systems with radial symmetry, with $R_a = 100$ nm and $R_a = 30$ nm.

Table 7

Values of the coefficients a_i , b_i , d_i , e_i , f_i and g_i for a $\text{Si}_{1-c}\text{Ge}_c$ alloy, $L = 100$ nm.

a_1	a_2	b_1	b_2
-0.00015	-1.5×10^{-6}	-0.000635	-1.15×10^{-6}
d_1	d_2	e_1	e_2
0.000587	1.21×10^{-6}	-0.057375	0.00003445
f_1	f_2	g_1	g_2
-0.134105	0.00006235	0.1341	-0.000062

Table 8

Values of the coefficients a_i , b_i , d_i , e_i , f_i and g_i for a $\text{Si}_c\text{Ge}_{1-c}$ alloy, $L = 30$ nm.

a_1	a_2	b_1	b_2
-0.0003	7×10^{-7}	-0.000825	7.5×10^{-7}
d_1	d_2	e_1	e_2
0.0012	-2×10^{-6}	-0.0123	0.000014
f_1	f_2	g_1	g_2
-0.20915	0.0004865	0.20785	-0.0004825

Table 9

Values of the coefficients a_i , b_i , d_i , e_i , f_i and g_i for a $\text{Si}_{1-c}\text{Ge}_c$ alloy, $L = 30$ nm.

a_1	a_2	b_1	b_2
0.00025	-1.5×10^{-6}	-0.000825	7.5×10^{-7}
d_1	d_2	e_1	e_2
0.00045	5×10^{-7}	-0.04905	0.0001125
f_1	f_2	g_1	g_2
-0.2092	0.000487	0.00586545	0.00155535

3. Rectification of the heat flux in nanowires

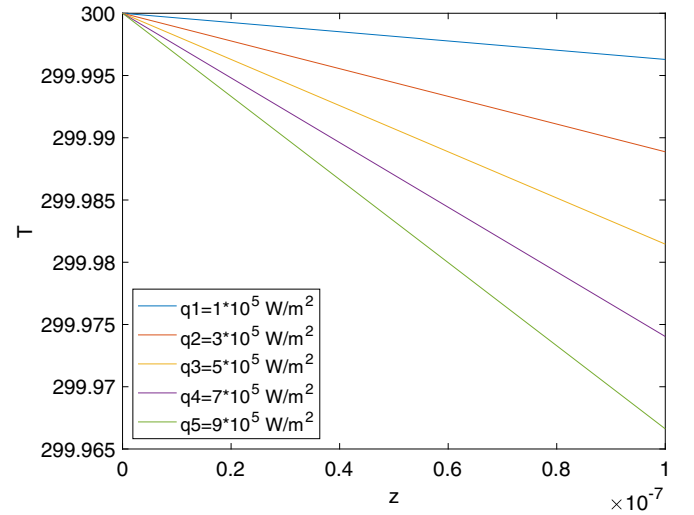
3.1. Temperature profiles

In the present section we carry out a computational analysis of heat transport along the direction z in $\text{Si}_c\text{Ge}_{1-c}$ nanowires (direct heat flux) and along $\text{Si}_{1-c}\text{Ge}_c$ nanowires (reverse heat flux), from $c = 0$ at $z = 0$ to $c = 1$ at $z = L$ in order to explore the consequences of several physical factors on the value of the rectification coefficient $R[c(z), T_H, T_C]$ going beyond the simplified situations analyzed in Refs. [4,20]. In fact, we explore the influence on R of the constitutive equation (1) for the thermal conductivity, of the spatial composition profile $c(z)$, of the value of the direct heat flux q_d , and of the length of the system L . To obtain the rectification coefficient, we must calculate the temperature profiles in the direct situation (heat flux from Ge to Si or, equivalently, heat flux in $\text{Si}_c\text{Ge}_{1-c}$) and in the reverse situation (heat flux from Si to Ge or, equivalently, heat flux in $\text{Si}_{1-c}\text{Ge}_c$), by:

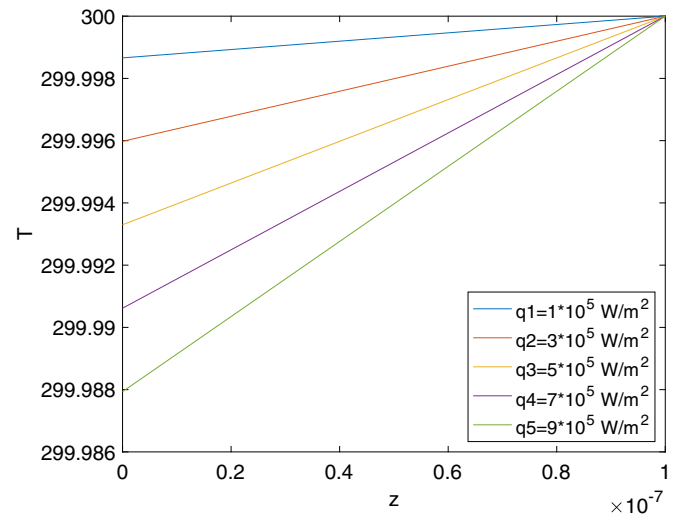
- (1) applying T_H (hottest T) and a given q_d on the Ge side, and obtaining the corresponding lowest temperature T_C at the Si side (direct situation);
- (2) applying T_H on the Si side and trying for different values of q_r until finding the value q_r for which the temperature at the Ge side is the lowest temperature T_C previously obtained for q_d (reverse situation).

The temperature profiles are obtained in steady state, under the hypothesis of validity of the Fourier law

$$q = -\lambda(T, c) \frac{dT}{dz}, \quad (5)$$



(a) From Ge to Si (direct flux)



(b) From Si to Ge (reverse flux)

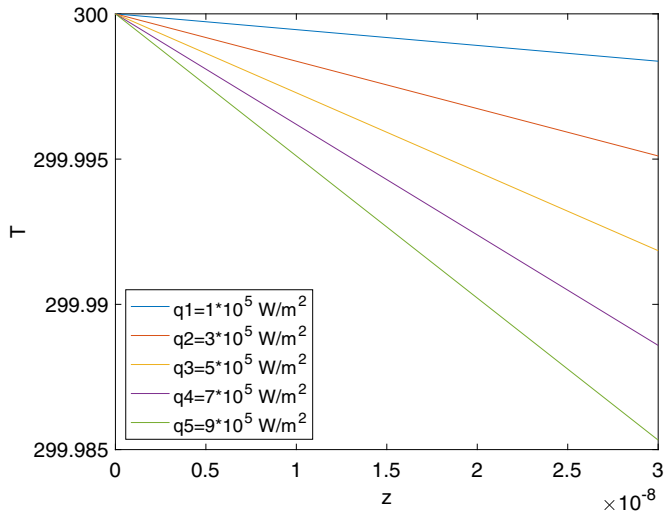
Fig. 2. Temperature profile for a linear composition profile $c(z) = z/L$, from $c = 0$ to $c = 1$, for q from $1 \times 10^5 \text{ W/m}^2$ to $9 \times 10^5 \text{ W/m}^2$ (as indicated in the figure) and $L = 100$ nm.

with a superimposed heat flux q_d , which will be constant along the one-dimensional system in steady state, because of the condition $\nabla \cdot \mathbf{q} = 0$. In this situation, we obtain the temperature profile by solving the differential equation

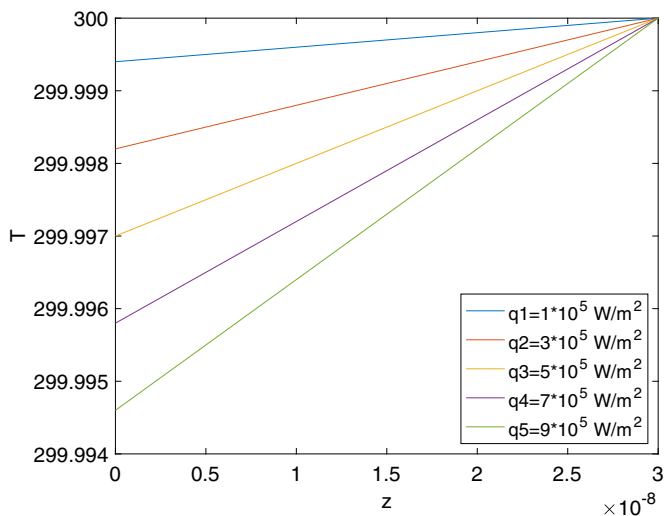
$$\frac{dT}{dz} = -\frac{q}{\lambda(T(z), c(z))}, \quad (6)$$

with $\lambda(T(z), c(z))$ given by Eq. (1) and with given composition profile $c(z)$.

In the direct situation $\text{Si}_c\text{Ge}_{1-c}$, we have considered that Ge ($c = 0$) is at the hot side ($z = 0$) (at 300 K) and Si ($c = 1$) is at the cold side ($z = L$). In the reverse situation $\text{Si}_{1-c}\text{Ge}_c$, instead, we have considered that Si ($c = 0$) is at the hot side ($z = 0$) and Ge ($c = 1$) is at the cold side ($z = L$). Indeed, we have assumed that the hot side is always at the left ($z = 0$), and that in the direct case pure Ge is at $z = 0$, while in the reverse case pure Si is at $z = 0$. In Fig. 2 we plot the temperature profiles corresponding to different values of the heat flux, for heat going from Ge to Si (direct case) Fig. 2(a), or from Si to Ge (reverse case) Fig. 2(b), always with $T = 300$ K



(a) From Ge to Si (direct flux)



(b) From Si to Ge (reverse flux)

Fig. 3. Temperature profile for a linear composition profile $c(z) = z/L$, from $c = 0$ to $c = 1$, for q from $1 \times 10^5 \text{ Wm}^{-2}$ to $9 \times 10^5 \text{ Wm}^{-2}$ (as indicated in the figure) and $L = 30 \text{ nm}$.

at the hottest side (left side, $z = 0$) and for a linear composition profile, namely $c(z) = z/L$, with $z \in [0, L]$, and $L = 100 \text{ nm}$. On the right-hand side of the figures are shown the values of T_C corresponding to different values of the direct heat flux. In Fig. 3 we plot the temperature profiles corresponding to different values of the heat flux, for heat going from Ge to Si (direct case) Fig. 3(a), or from Si to Ge (reverse case) Fig. 3(b), always with $T = 300 \text{ K}$ at the hottest side (left side, $z = 0$) and for a linear composition profile, namely $c(z) = z/L$, with $z \in [0, L]$, and $L = 30 \text{ nm}$. Nonlinear composition profiles, namely $c(z) = (z/L)^2$, $c(z) = (z/L)^3$, could also be considered by using the same method as above. In future works, we intend to analyze and compare these cases with the linear composition profile.

3.2. Rectification coefficient

Once the temperature profiles have been obtained, the rectification coefficient can be calculated by the mathematical procedure illustrated by the items 1) and 2) above. In Tables 10 and 11,

Table 10

Rectification coefficient for linear composition profiles, $c(z) = z/L$, for different values of the direct heat flux from $1 \times 10^5 \text{ Wm}^{-2}$ to $9 \times 10^5 \text{ Wm}^{-2}$. The T_H and T_C highest and lowest boundary temperatures are also indicated and $L = 100 \text{ nm}$.

q_d	R	T_H	T_C
1×10^5	$2.77/1 = 2.7700$	300	299.9963
3×10^5	$8.30/3 = 2.7667$	300	299.9889
5×10^5	$13.8/5 = 2.7600$	300	299.9815
7×10^5	$19.365/7 = 2.7664$	300	299.9740
9×10^5	$24.9/9 = 2.7667$	300	299.9666

Table 11

Rectification coefficient for linear composition profiles, $c(z) = z/L$, for different values of the direct heat flux from $1 \times 10^5 \text{ Wm}^{-2}$ to $9 \times 10^5 \text{ Wm}^{-2}$. The T_H and T_C highest and lowest boundary temperatures are also indicated and $L = 30 \text{ nm}$.

q_d	R	T_H	T_C
1×10^5	$2.718/1 = 2.7180$	300	299.9984
3×10^5	$8.15/3 = 2.7167$	300	299.9951
5×10^5	$13.6/5 = 2.7200$	300	299.9918
7×10^5	$19/7 = 2.7143$	300	299.9886
9×10^5	$24.46/9 = 2.7178$	300	299.9853

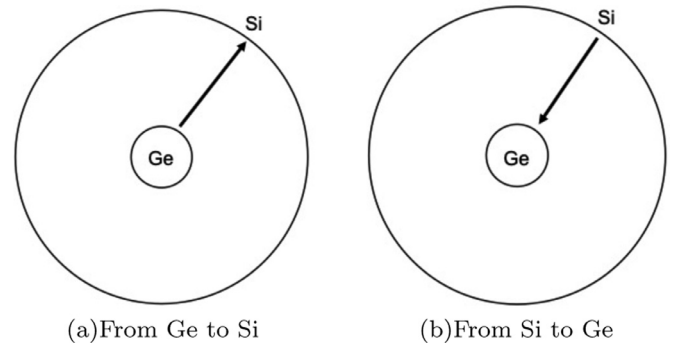


Fig. 4. Direct and reverse heat flow for Ge inside and Si outside.

we show the corresponding values of the rectification coefficient for linear composition profile $c = z/L$, for different values of the direct heat flux from $1 \times 10^5 \text{ Wm}^{-2}$ to $9 \times 10^5 \text{ Wm}^{-2}$, both for $L = 100 \text{ nm}$ and $L = 30 \text{ nm}$. It can be seen that R is almost independent of the intensity of the direct heat flux, and is a little bit higher for $L = 100 \text{ nm}$. However, we observe that R is very high, and this result confirms the good performance of graded materials as thermal rectifiers.

4. Rectification in cylindrical nanodevices

4.1. Fourier law in cylindrical coordinates

In this section we consider a cylindrical system with axial symmetry, undergoing radial heat flow along the direction r , and $r \in [r_{int}, r_{ext}]$ with r_{int} and r_{ext} being the internal and external radii. We combine such a geometry with graded $\text{Si}_c\text{Ge}_{1-c}$ composition, where c depends linearly on r , namely $c(r) = \frac{r - r_{int}}{r_{ext} - r_{int}}$. We will consider the direct and reverse heat flow in the following situations:

- Ge inside Si outside (Fig. 4);
- Si inside Ge outside (Fig. 5).

We will denote as direct heat flux q_d the outward one, and as reverse heat flux q_r the inward one. In these situations the differ-

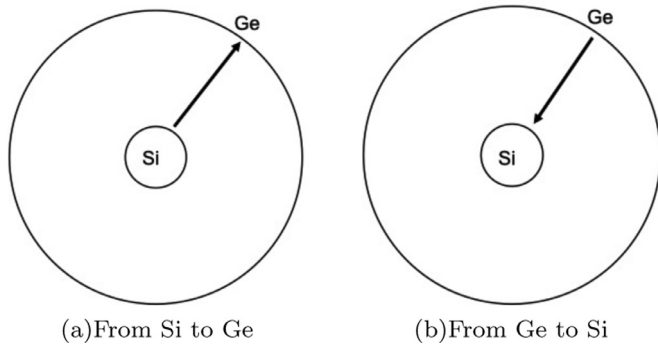


Fig. 5. Direct and reverse heat flow for Si inside and Ge outside.

Table 12
Rectification coefficient for linear profile $c(r) = \frac{r - r_{int}}{r_{ext} - r_{int}}$, for different values of the direct heat flux in the range of 10^9Wm^{-2} , for $r_{int} = 50 \text{ nm}$ and $r_{ext} = 150 \text{ nm}$.

Γ	R	T_H	T_C
2×10^2	$2.82272/2 = 1.4114$	300	250.5158
4×10^2	$5.642145/4 = 1.4105$	300	201.3330
6×10^2	$8.460195/6 = 1.4100$	300	152.3789
8×10^2	$11.278775/8 = 1.4098$	300	103.5833

ence in the value of the internal and external radii acts as a further inhomogeneity of geometrical kind, which may enhance or reduce the effects of the material inhomogeneity in the concentration. Indeed, in the heat flow along a nanowire, the two situations described above (Ge inside, Ge outside) would be equivalent, since they would correspond to having pure Ge on the left hand side or on the right hand side of the nanowire, and it would be inconsequential for the value of the rectification. In the axial case, instead, they become physically differentiated, because of the geometry (the area corresponding to Ge in the cases Ge inside and Ge outside is no longer the same one).

In this geometry, the Fourier law takes the form

$$q = \frac{\Gamma}{r} = -\lambda(T(r), c(r)) \frac{dT}{dr}, \tag{7}$$

where Γ is a constant (proportional to the total heat per unit time supplied to the system per unit length of the cylinder). The form of q in (7) follows by the equation $\nabla \cdot \mathbf{q} = 0$ (steady state condition) which, in polar coordinates and in the presence of radial symmetry, leads to $q(r) \sim 1/r$. Thus, the temperature profile will be found by solving the differential equation

$$\frac{dT}{dr} = -\frac{\Gamma}{\lambda(T(r), c(r))r}. \tag{8}$$

We take Γ of order of magnitude 10^2 Wm^{-1} , in order to have a heat flux of order of magnitude 10^9 Wm^{-2} for $r \in [100 \text{ nm}, 150 \text{ nm}]$, and a heat flux of order of magnitude 10^{10} Wm^{-2} for $r \in [10 \text{ nm}, 40 \text{ nm}]$.

4.2. Rectification coefficient in cylindrical geometry

In Table 12, we give the values of the rectification coefficient for $c(r) = \frac{r - r_{int}}{r_{ext} - r_{int}}$, with $r_{int} = 50 \text{ nm}$ and $r_{ext} = 150 \text{ nm}$ (from Ge to Si direct case, from Si to Ge reverse case).

In Table 13, we give the values of the rectification coefficient for $c(r) = \frac{r - r_{int}}{r_{ext} - r_{int}}$, with $r_{int} = 50 \text{ nm}$ and $r_{ext} = 150 \text{ nm}$ (from Si to Ge direct case, from Ge to Si reverse case).

Table 13
Rectification coefficient for linear profile $c(r) = \frac{r - r_{int}}{r_{ext} - r_{int}}$, for different values of the direct heat flux in the range of 10^9Wm^{-2} , for $r_{int} = 50 \text{ nm}$ and $r_{ext} = 150 \text{ nm}$.

Γ	R	T_H	T_C
2×10^2	$3.8011/2 = 1.9005$	300	288.6905
4×10^2	$7.60195/4 = 1.9005$	300	277.3863
6×10^2	$11.4029/6 = 1.9005$	300	266.0857
8×10^2	$15.2043/8 = 1.9005$	300	254.7868

Table 14
Rectification coefficient for linear profile $c(r) = \frac{r - r_{int}}{r_{ext} - r_{int}}$, for different values of the direct heat flux in the range of 10^{10}Wm^{-2} , for $r_{int} = 10 \text{ nm}$ and $r_{ext} = 40 \text{ nm}$.

Γ	R	T_H	T_C
2×10^2	$2.77405/2 = 1.3870$	300	263.0657
4×10^2	$5.536585/4 = 1.3841$	300	226.7908
6×10^2	$8.28525/6 = 1.3809$	300	191.2387
8×10^2	$11.01813/8 = 1.3773$	300	156.4585

Table 15
Rectification coefficient for linear profile $c(r) = \frac{r - r_{int}}{r_{ext} - r_{int}}$, for different values of the direct heat flux in the range of 10^{10}Wm^{-2} , for $r_{int} = 10 \text{ nm}$ and $r_{ext} = 40 \text{ nm}$.

Γ	R	T_H	T_C
2×10^2	$3.46703/2 = 1.7335$	300	290.5539
4×10^2	$6.93557/4 = 1.7339$	300	281.0937
6×10^2	$10.40599/6 = 1.7343$	300	271.6178
8×10^2	$13.87859/8 = 1.7348$	300	262.1244

In Table 14, we give the values of the rectification coefficient for $c(r) = \frac{r - r_{int}}{r_{ext} - r_{int}}$, with $r_{int} = 10 \text{ nm}$ and $r_{ext} = 40 \text{ nm}$ (from Ge to Si direct case, from Si to Ge reverse case).

In Table 15, we give the values of the rectification coefficient for $c(r) = \frac{r - r_{int}}{r_{ext} - r_{int}}$, with $r_{int} = 10 \text{ nm}$ and $r_{ext} = 40 \text{ nm}$, (from Si to Ge direct case, from Ge to Si reverse case).

Although smaller with respect to that of the linear case, the rectification coefficient corresponding to the cylindrical case is also very high and independent of the intensity of the applied heat flux. Moreover, also in this case, it seems to increase with the length of the system.

5. Concluding remarks

In the present paper the analysis developed in Ref. [5] has been applied by using a different constitutive equation for the thermal conductivity, i.e., Eq. (1), which has been obtained by determining a mathematical fit of the experimental results [21]. It is seen that the value of R in the full range $0 \leq c \leq 1$ is higher than 2.7 (Tables 10 and 11), thus indicating a meaningful rectification, both for nanowires of length $L = 100 \text{ nm}$, and of length $L = 30 \text{ nm}$. For cylindrical systems with transversal section of radius $r_{int} = 50 \text{ nm}$ and $r_{ext} = 150 \text{ nm}$, we got $R \approx 1.41$ if we have the direct case from Ge to Si and the reverse case from Si to Ge, and $R \approx 1.90$ if we have the direct case from Si to Ge, and the reverse case from Ge to Si (Tables 12 and 13). For cylindrical systems with transversal section of radius $r_{int} = 10 \text{ nm}$ and $r_{ext} = 40 \text{ nm}$, we got $R \approx 1.38$ if we have the direct case from Ge to Si and the reverse case from Si to Ge, and $R \approx 1.73$ if we have the direct case from Si to Ge and the reverse case from Ge to Si (Tables 14 and 15). The results above

denote a meaningful improvement of the rectification coefficient, both for linear and cylindrical systems, with respect to the results obtained in [5]. In future researches we plan to analyze the same problem for the bulk, namely, for L in the interval [1 mm, 4 mm]. Indeed, for such a length the function in Eq. (1) does not seem to fit the experimental data with the same precision of nanometric scale and some improvement seems to be necessary. We also plan to explore composition spatial distribution non linear in the position (namely $c(z) = (z/L)^a$), with a an exponent and composition ranges from $c = 0$ to 0.1, and $c = 0.9$ to 1. Finally, we remark that the results above have been obtained under the hypothesis of validity of the classical Fourier's law. However, one can find in literature several theories of heat conduction beyond the Fourier's law [22–25]. For instance, in Extended Irreversible Thermodynamics [24,26,27], the heat flux is no longer assigned by a constitutive equation but is determined by a suitable balance law [24,26–29]. Thus, it would be interesting to explore, for different geometries, the influence on R of the additional term $\ell^2 \nabla^2 \mathbf{q}$ with ℓ as the mean free path of phonons in the expression of the heat flux. Moreover, for materials of the type considered here, it is interesting to investigate how ℓ changes with c , and determine a suitable law of dependence on c .

Declaration of Competing Interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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